

缸盖蠕铁高温服役性能与损伤机制评价表征

Service properties and damage mechanisms of compacted graphite iron
for cylinder head at high temperature

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- ◆ **一、缸盖研究背景与主要工程需求**
Research background and main engineering requirements
- ◆ **二、拉伸损伤机制与性能定量表征**
Tensile damage mechanism and property characterization
- ◆ **三、高温疲劳损伤机制与强度预测**
Fatigue damage mechanism and strength prediction
- ◆ **四、疲劳寿命预测模型与损伤机制**
Fatigue life prediction model and damage mechanism
- ◆ **五、主要结论**
Main conclusion

一、缸盖研究背景与主要工程问题

Research background and main engineering question

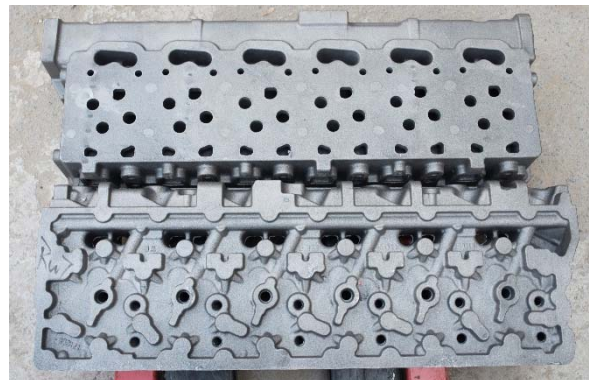
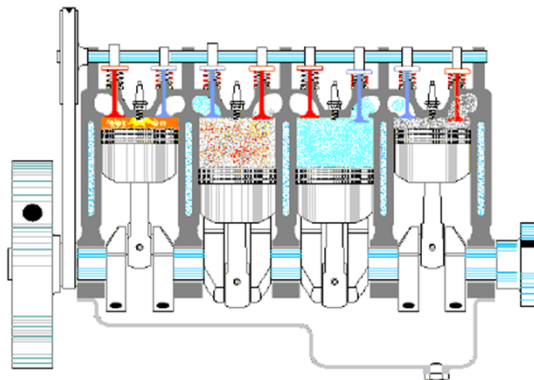
1. cylinder head for diesel engine



Application of diesel engine



Service condition of cylinder head



High-Temperature

High-Pressure

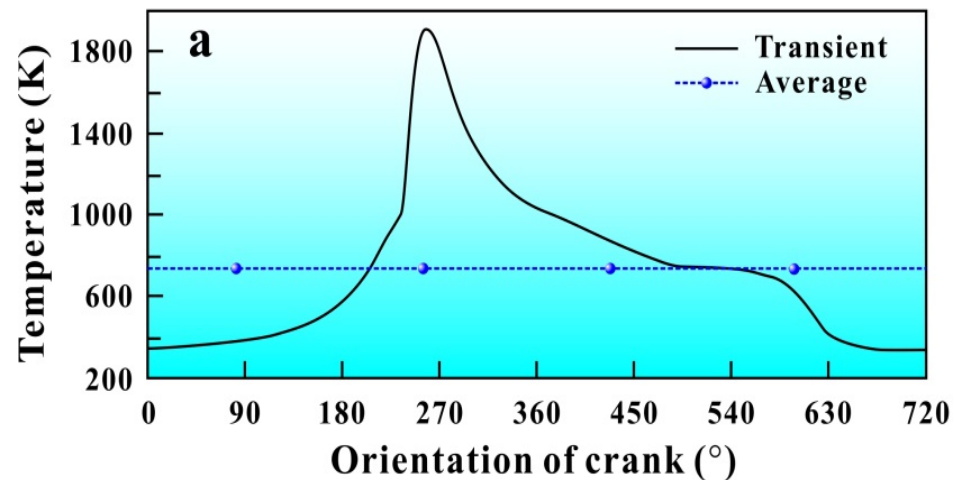
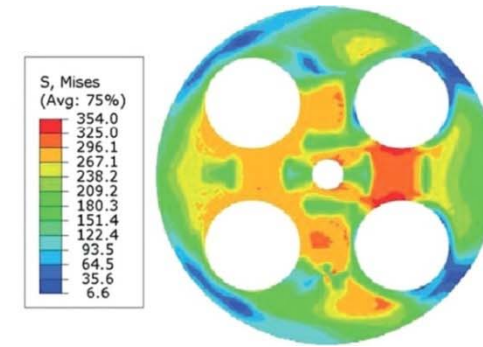
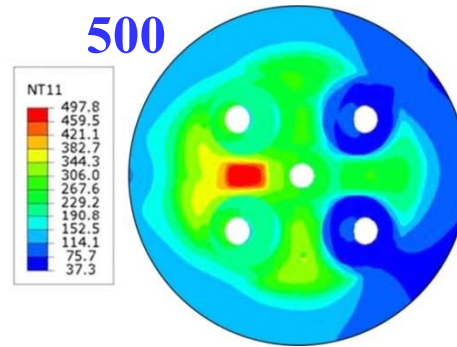
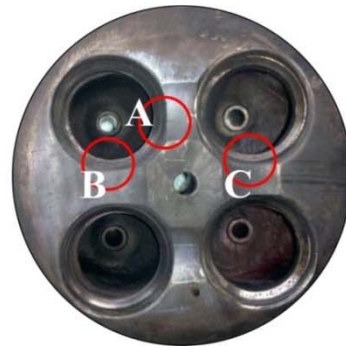
High-Speed

High-Efficient

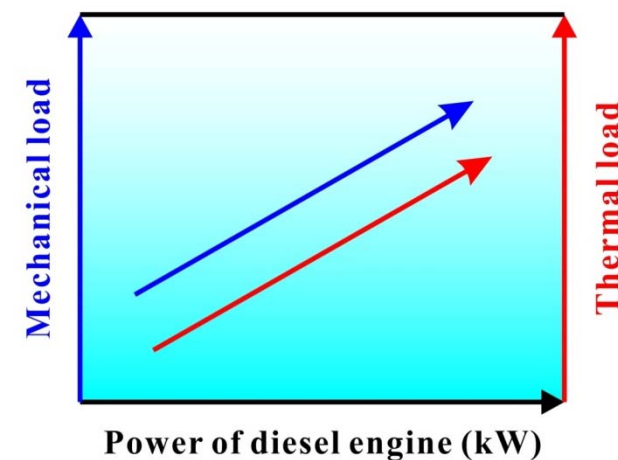
High reliability Service properties at high temperature

2. Service component and material requirement

Cylinder
head
(CGI)



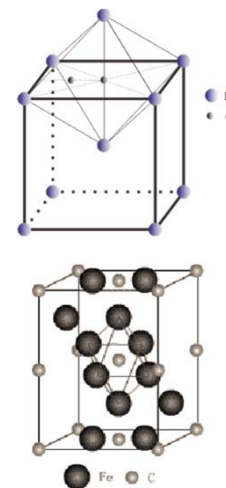
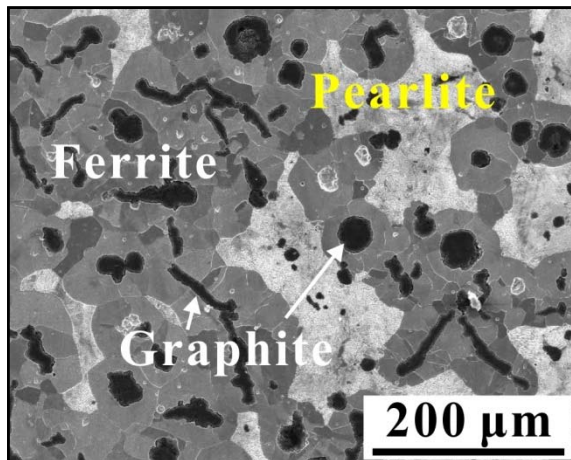
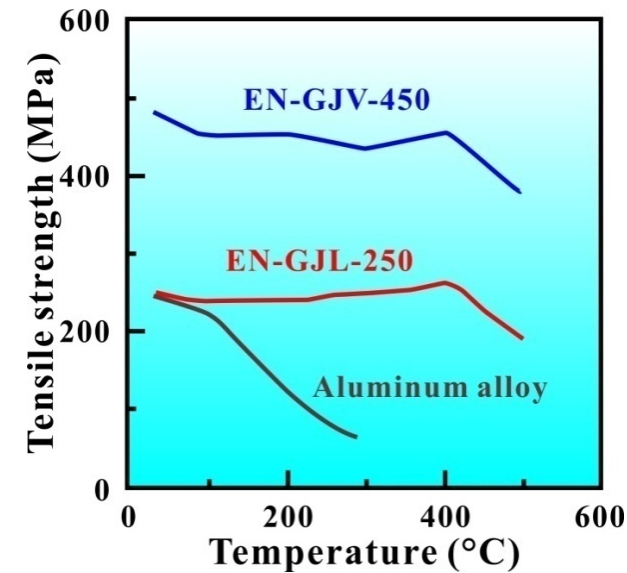
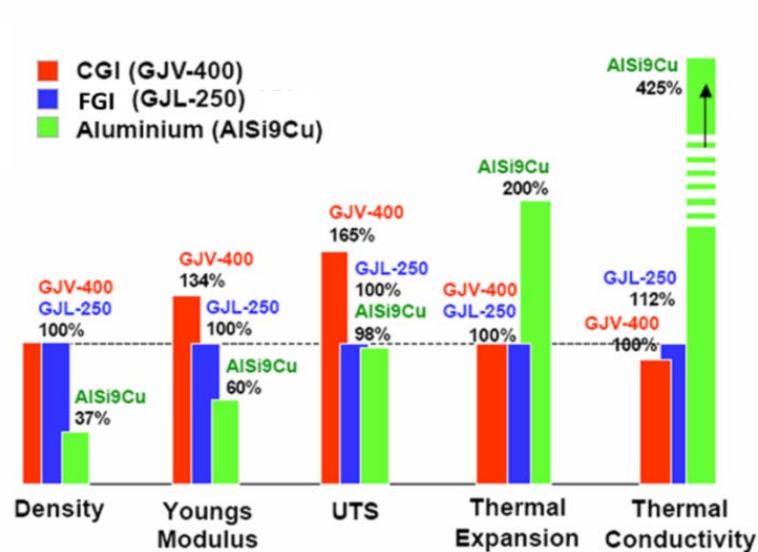
缸内燃气温度



功率和机械热载荷的关系

Complex harsh condition: Load & temperature

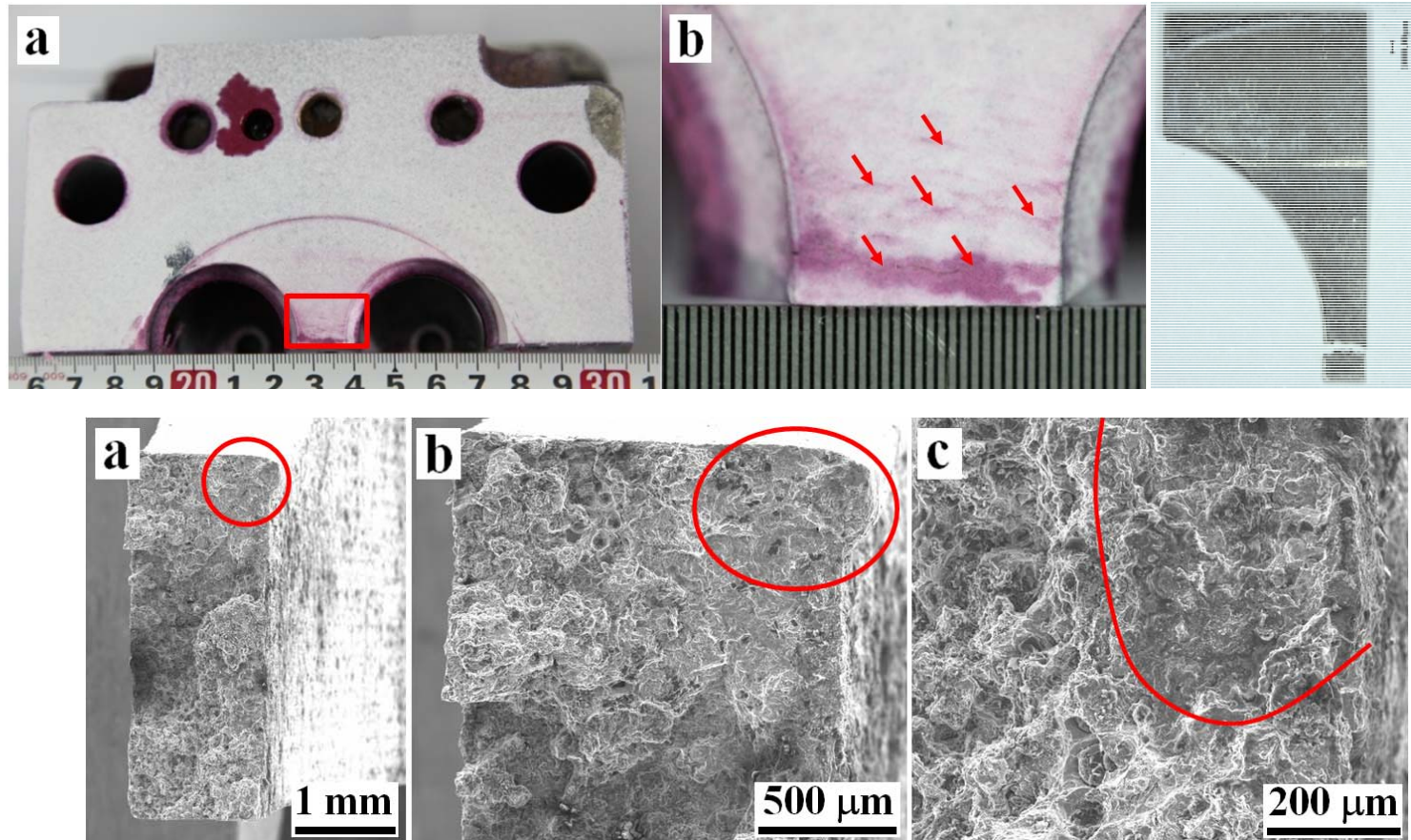
3. Material and microstructure



Elongation: ~ 50% **F**
Strength: 180~280 MPa
Hardness: HB 50~80

Elongation : ~ 0 **Fe₃C**
Strength : 35 MPa
Hardness : HB 800

4. Failure mode at service conditions



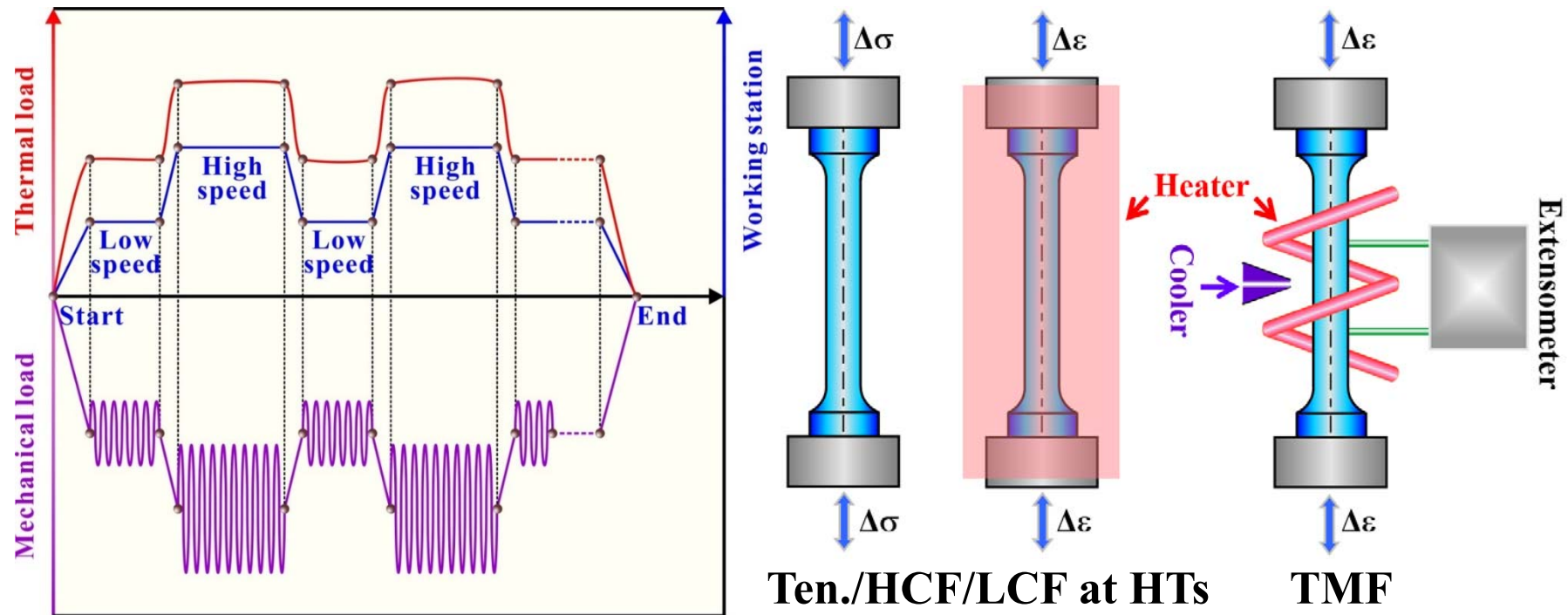
Fatigue & oxidation

More failure less fatigue

4. Failure mode at service conditions

热机疲劳：温度和机械载荷按一定相位变化

TMF: Temperature and mechanical load change with one phase

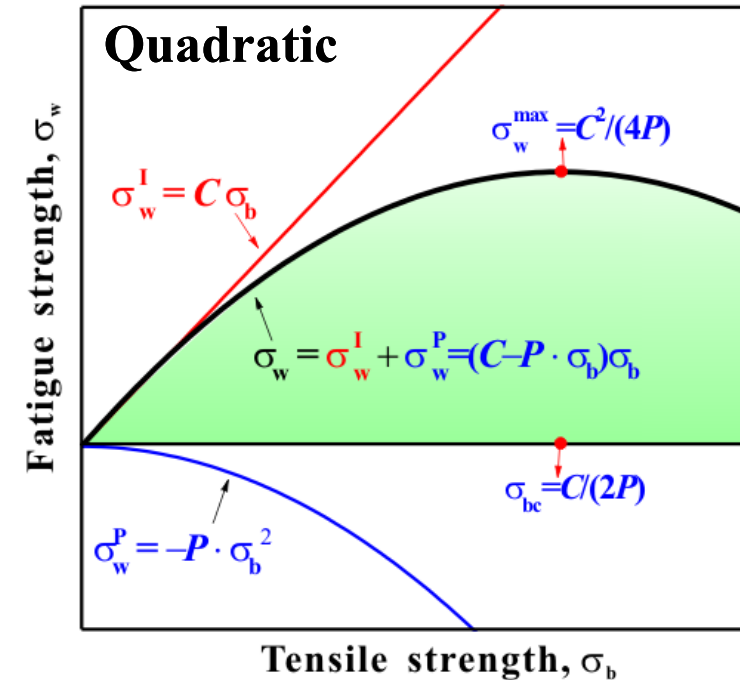
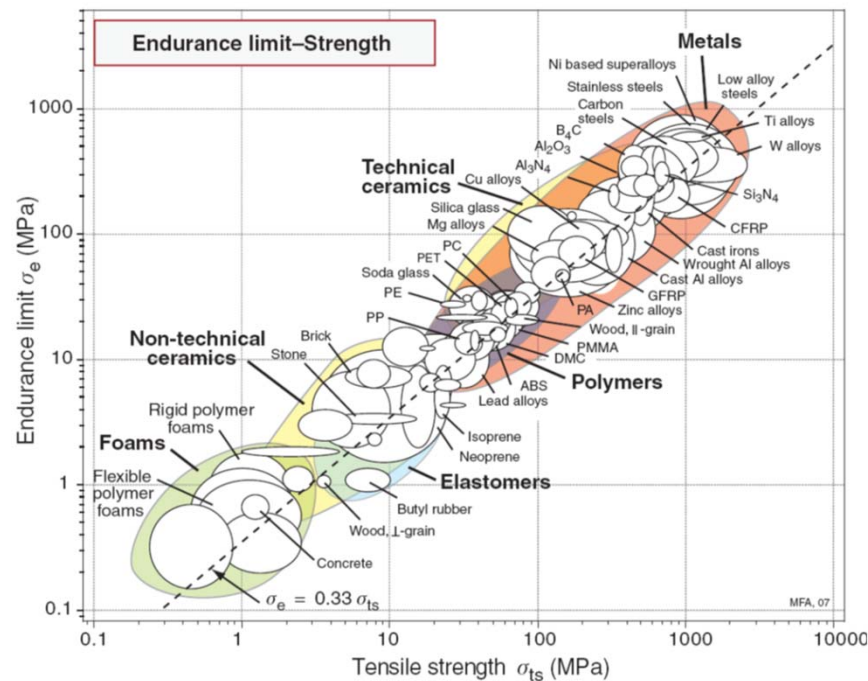


Sketch of load spectrum for cylinder head

Heating and loading

Tensile at HTs	HCF at HTs	LCF at HTs	TMF
Tensile strength	HCF strength	LCF life	TMF life

5. Question of service property prediction



1870年

$$\sigma_w = m \sigma_b$$

Whöler

$m: 0.4 \sim 0.5$

Father of
fatigue

Linear
relation

$$\sigma_w = (C - P \times \sigma_b) \times \sigma_b$$

P : Process Defect

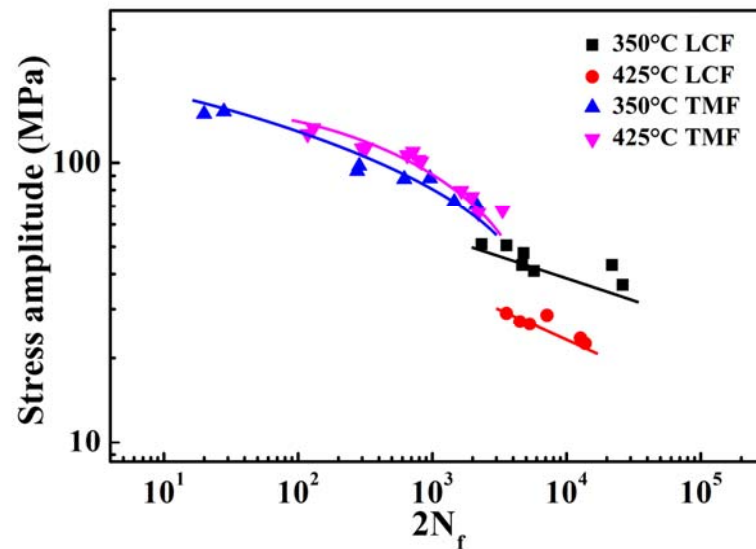
C : Intrinsic Defect

Materials Component

5. Question of service property prediction

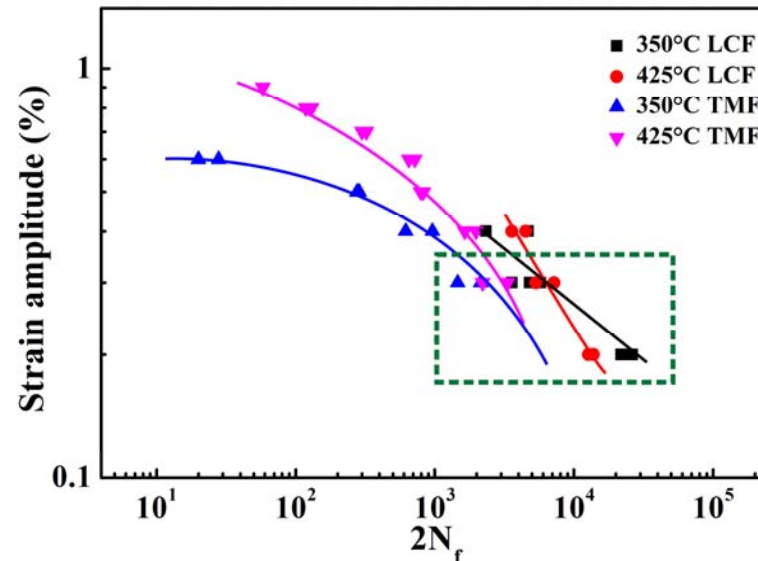
Basquin, 1910

$$\sigma_a = \sigma_f (2N_f)^b$$
$$D_i = \frac{1}{N_f} = 2 \cdot \left(\frac{\sigma_a}{\sigma_f} \right)^{-1/b}$$



Coffin & Manson, 1954

$$\varepsilon_p = \varepsilon_f (2N_f)^c$$
$$D_i = \frac{1}{N_f} = 2 \cdot \left(\frac{\varepsilon_p}{\varepsilon_f} \right)^{-1/c}$$



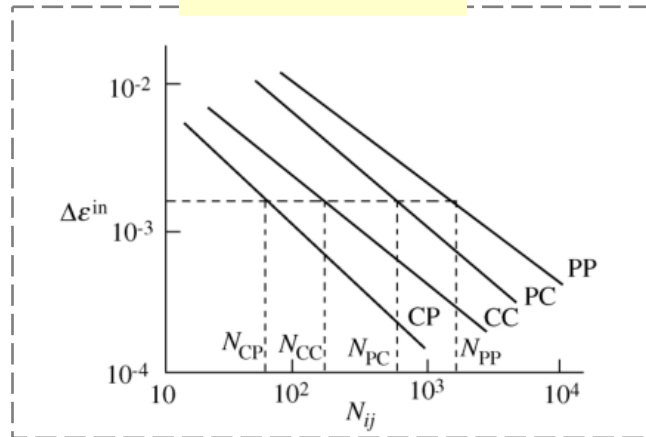
Thermal & force



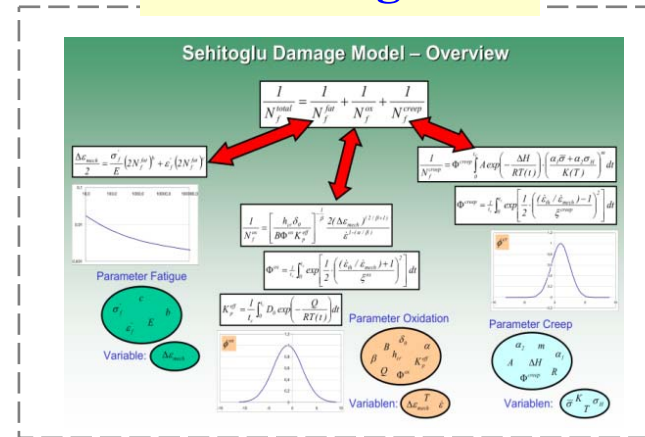
Typical model: Limited

5. Question of service property prediction

SRP



Sehitoglu



Ostergren

$$\Delta W_T = \sigma_{\max} \cdot \Delta \varepsilon_p$$

$$N_f^\beta (\Delta W - \Delta W_0)^m = C$$

$$\sigma_{\max} \cdot \Delta \varepsilon_p \cdot N_f^\beta \cdot \nu^{\beta(k-1)} = C$$

Miller

$$\frac{da}{dN} = \left. \frac{da}{dN} \right|_{\text{fatigue}} + \left. \frac{da}{dN} \right|_{\text{ox}} + \left. \frac{da}{dN} \right|_{\text{creep}}$$

$$\left. \frac{da}{dN} \right|_{\text{creep}} = C_c \hat{C}^{mc} \left. \frac{da}{dN} \right|_{\text{fatigue}} = C_f \Delta J^{mf}$$

$$\left. \frac{da}{dN} \right|_{\text{ox}} = C_0 \Delta J^{m_0} \Delta t^{(1/2-\xi)}$$

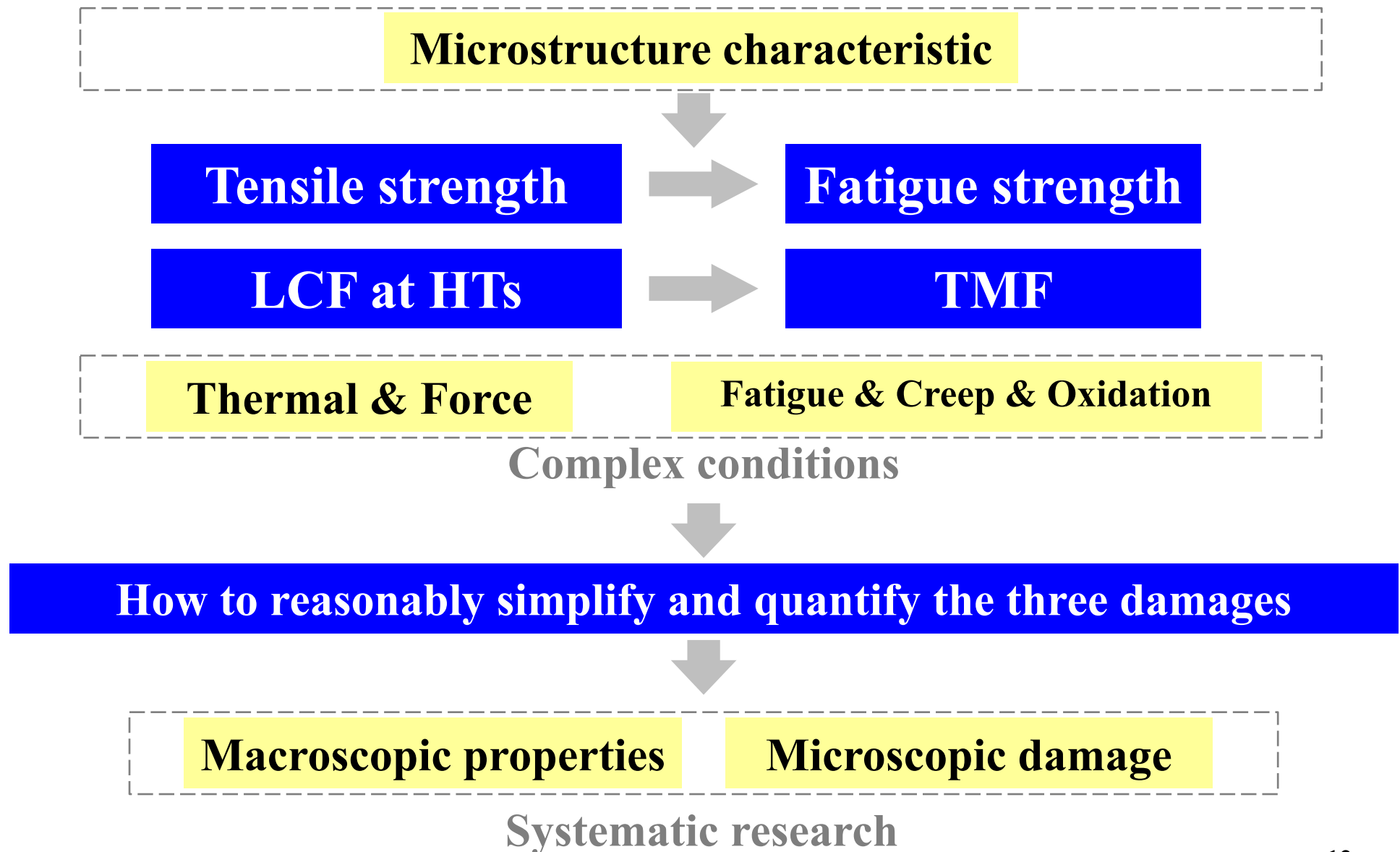
Thermal & Force



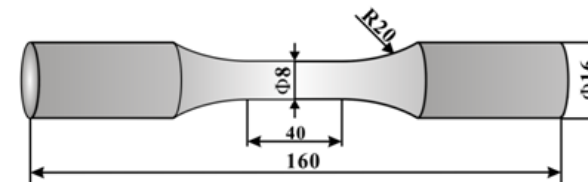
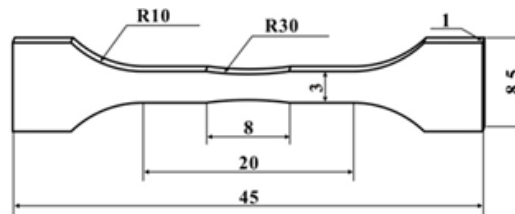
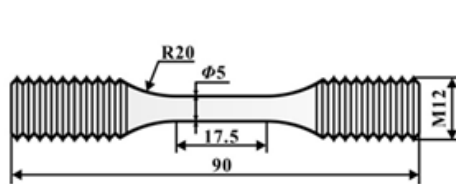
New model: Complex

Coffin LF, *et al.* *Proc. Inst. Mech. Eng.* 1974;188:109. Neu RW, Sehitoglu H, *Metall. Mater. Trans. A*, 1998;20: 1755. Miller MP, D.L. *et al.* *J. Eng. Mater. Technol.*, 1992;114: 2346.

6. Simplified and quantified service properties at HTs



Experimental methods



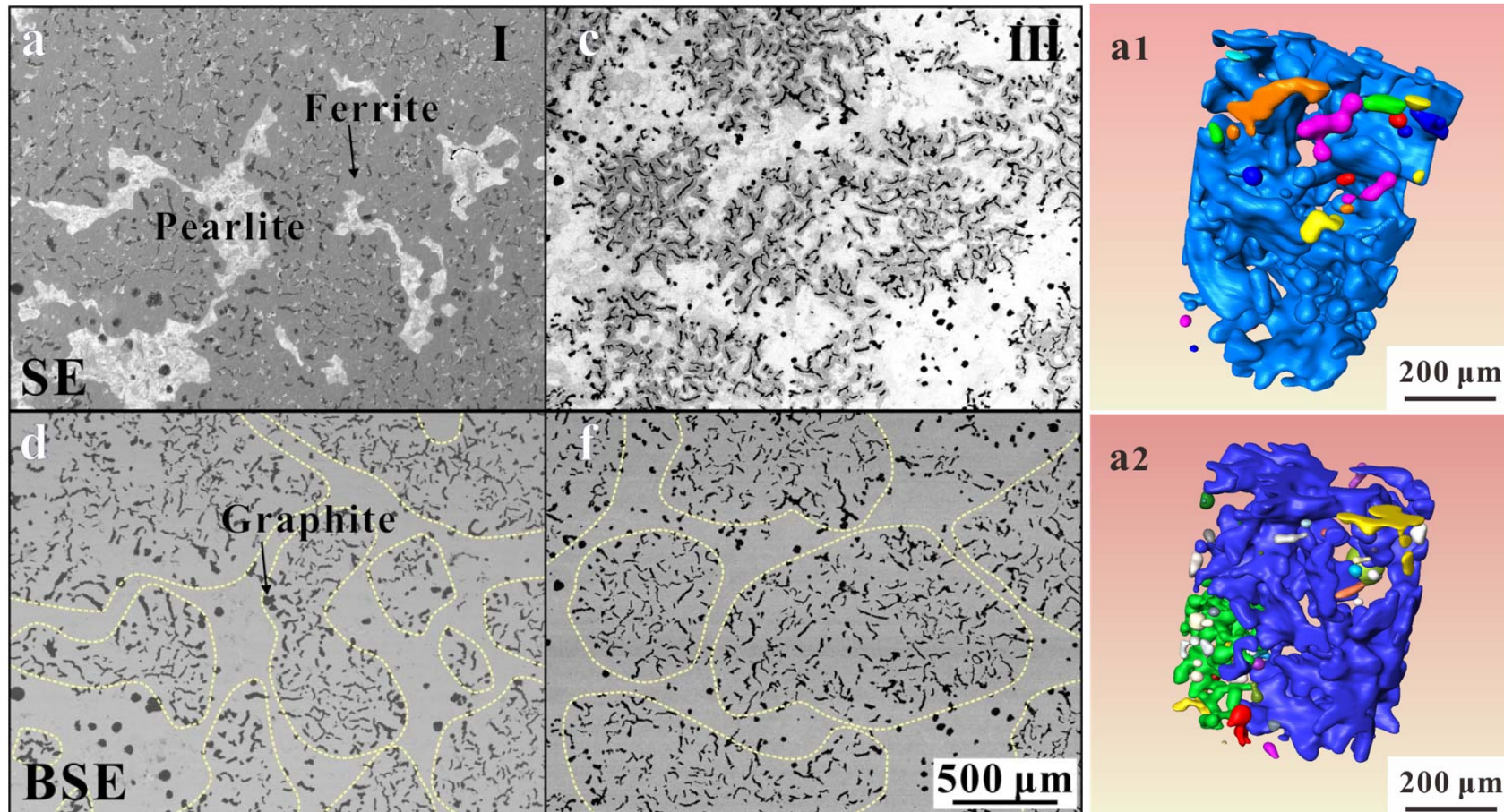
Ten./ LCF ratio : $10^{-2}/s \setminus 10^{-4}/s$

TMF Tem.: 25~400, 25~500 °C

FS: Staircase 5 Pairs 1×10^7

HCF/LCF Tem.: 25, 400, 500 °C

Microstructures of samples



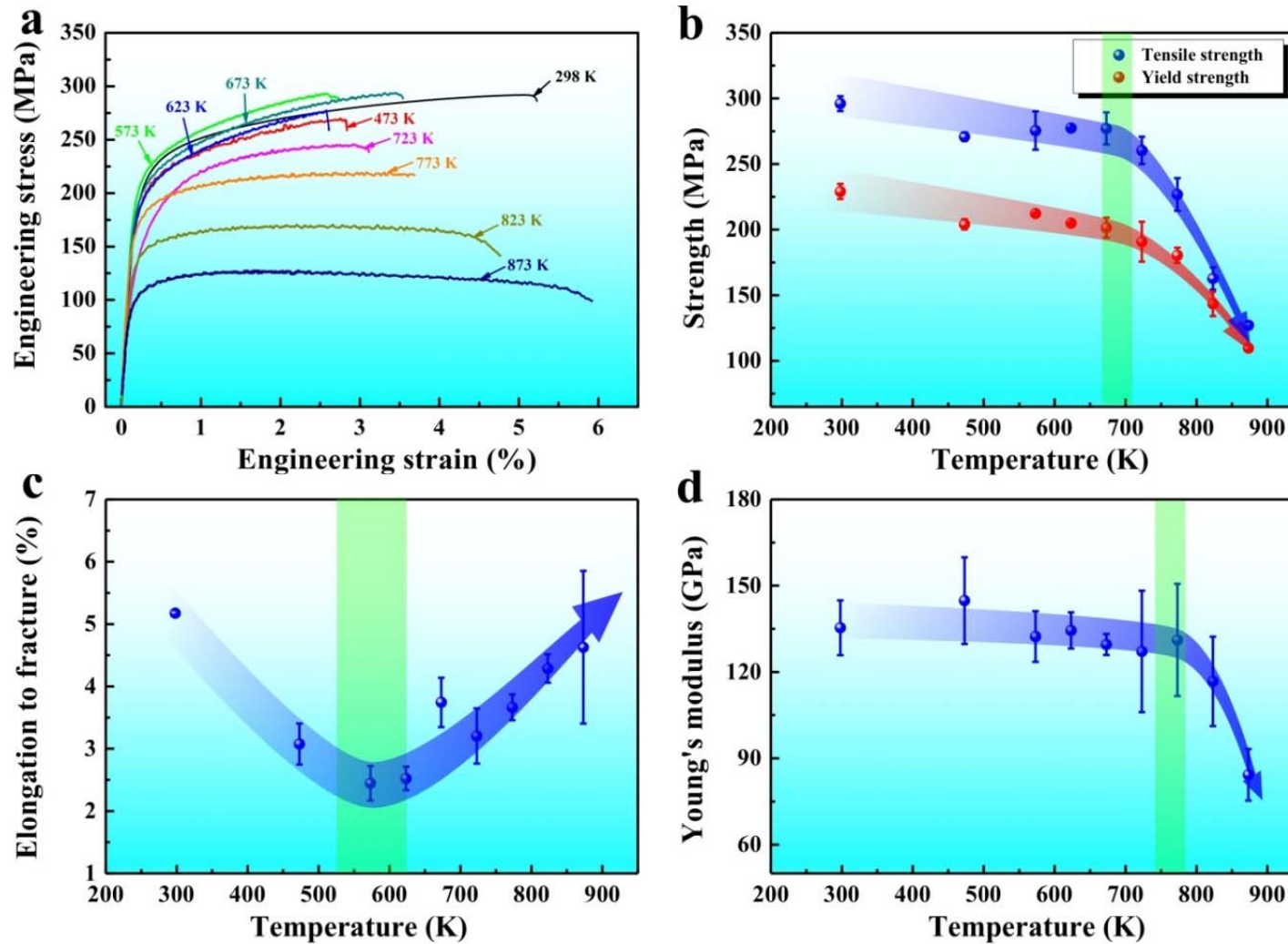
F:10%~60% P:15%~80% G:~10% VG:60%~90%

Recognition of traditional materials: **composite**

二、拉伸损伤机制与性能定量表征

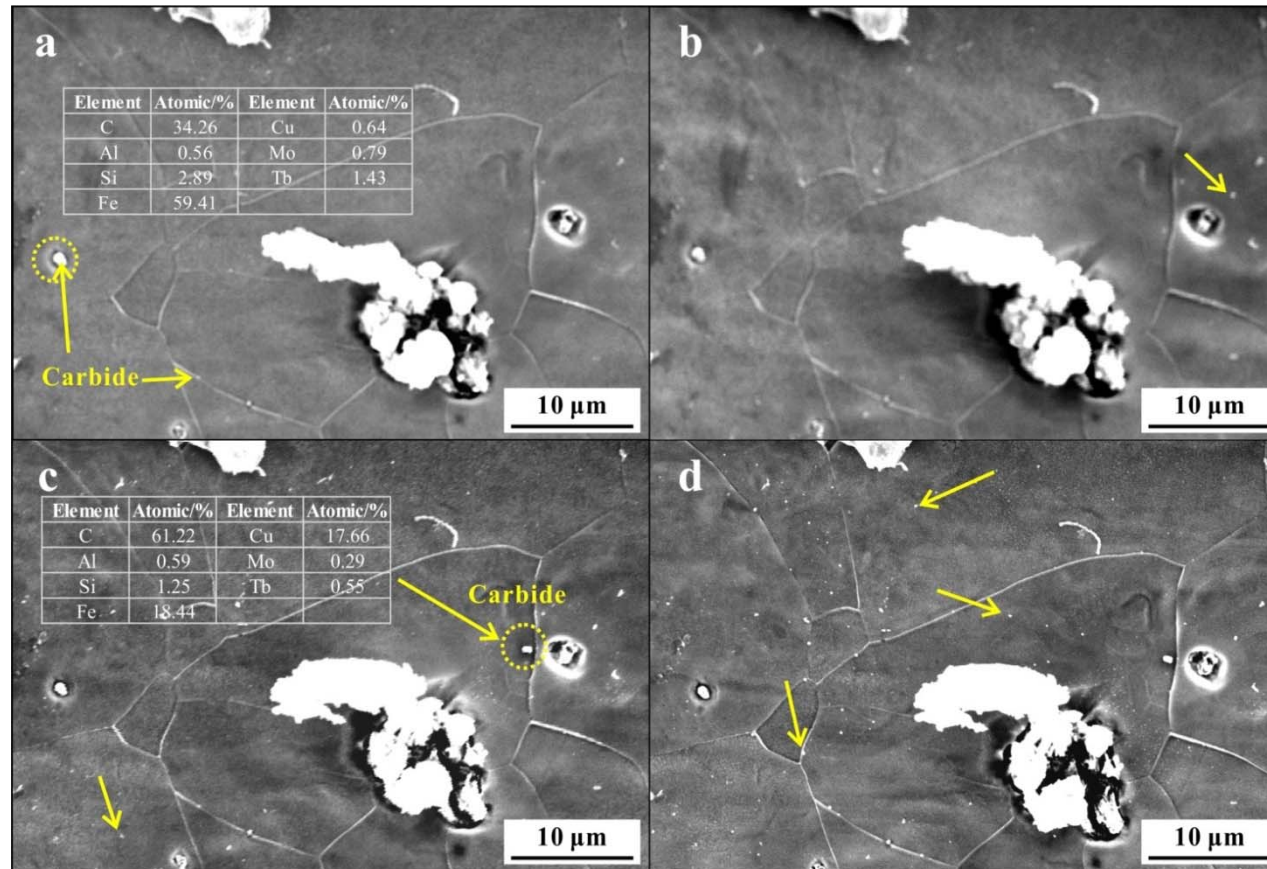
Tensile damage mechanism and property characterization

1. Tensile properties of CGI



2. Tensile mechanism of CGI

In situ ferrite evolution

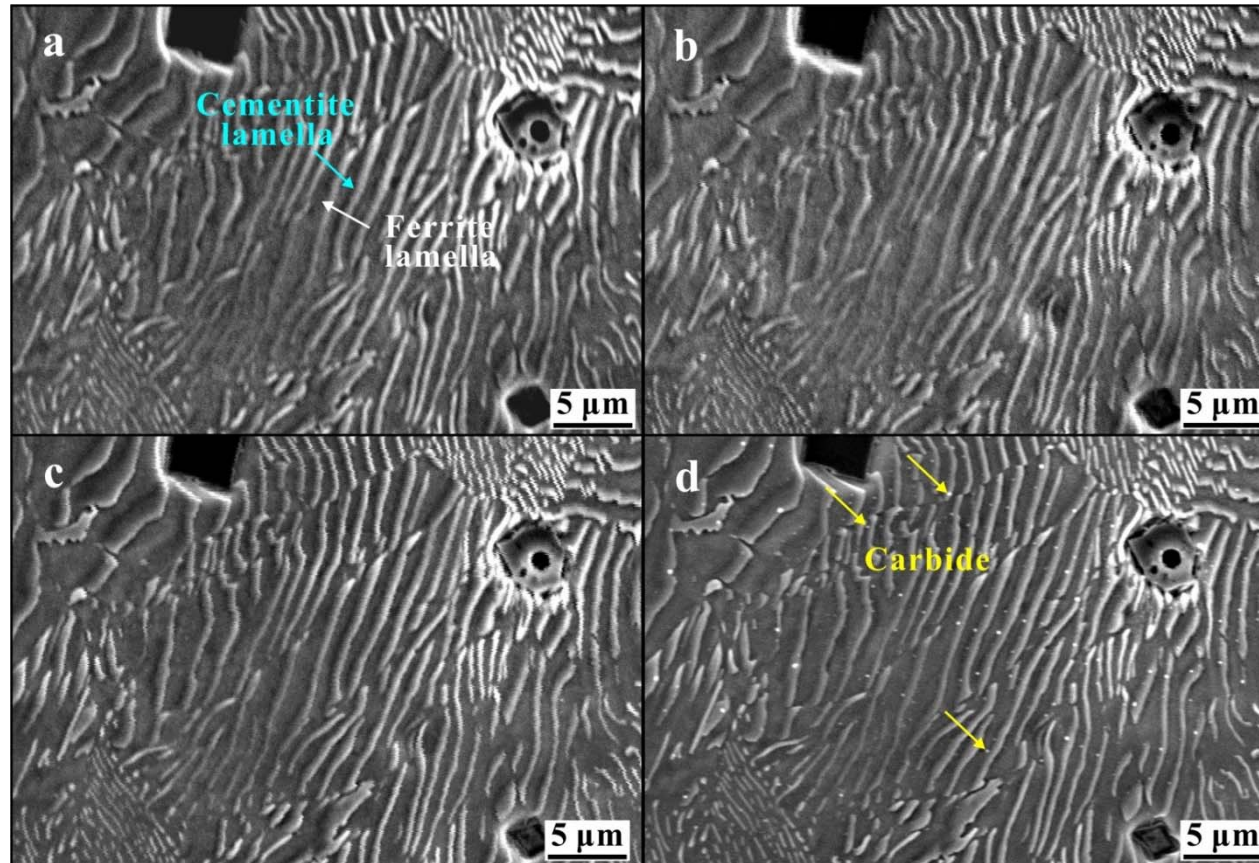


(a) 25°C; (b) 200°C; (c) 400°C; (d) 600°C

Carbon diffusion Carbide formation

2. Tensile mechanism of CGI

In situ pearlite evolution

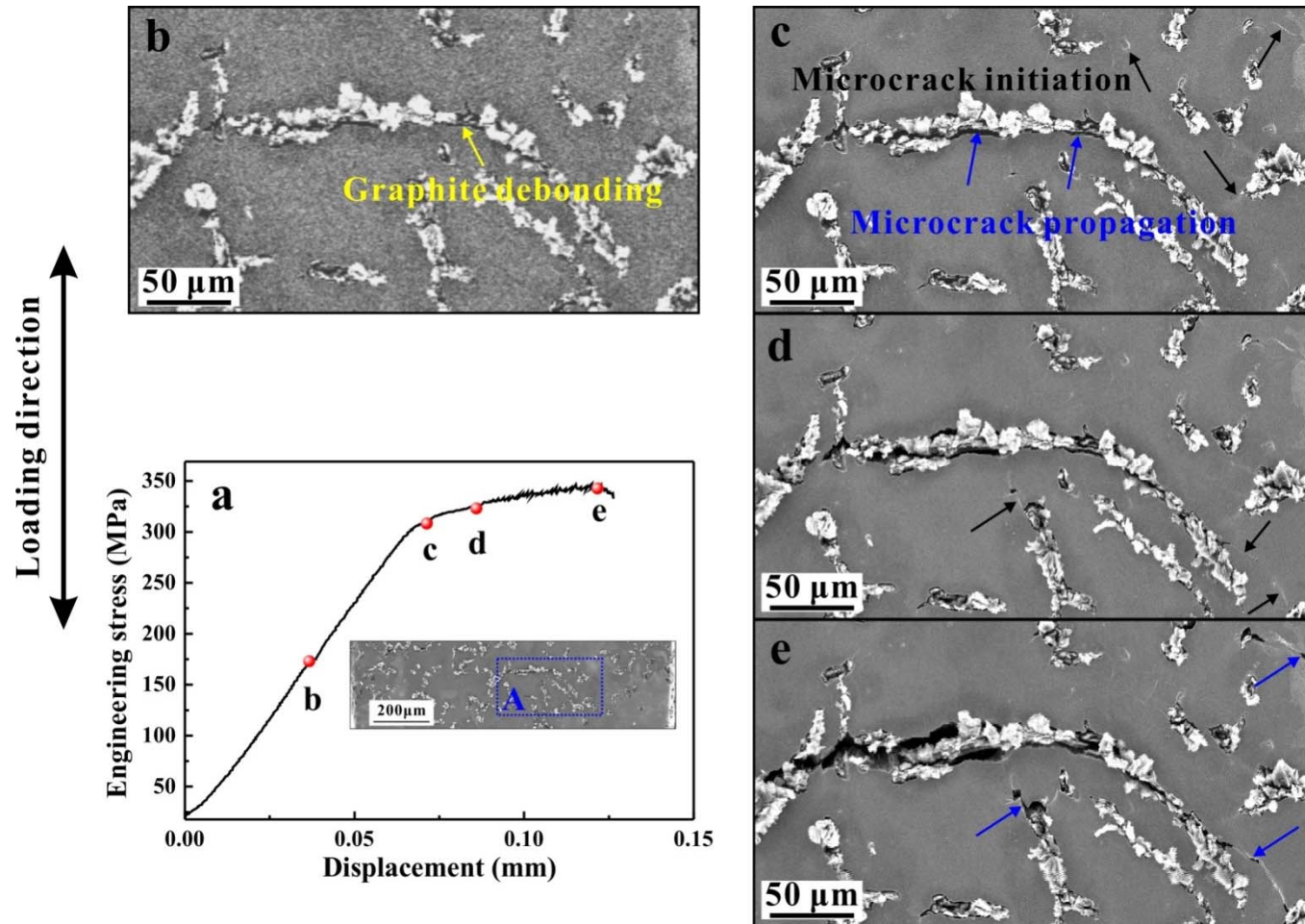


(a) 25°C; (b) 200°C; (c) 400°C; (d) 600°C

Carbon diffusion Carbide formation

2. Tensile mechanism of CGI

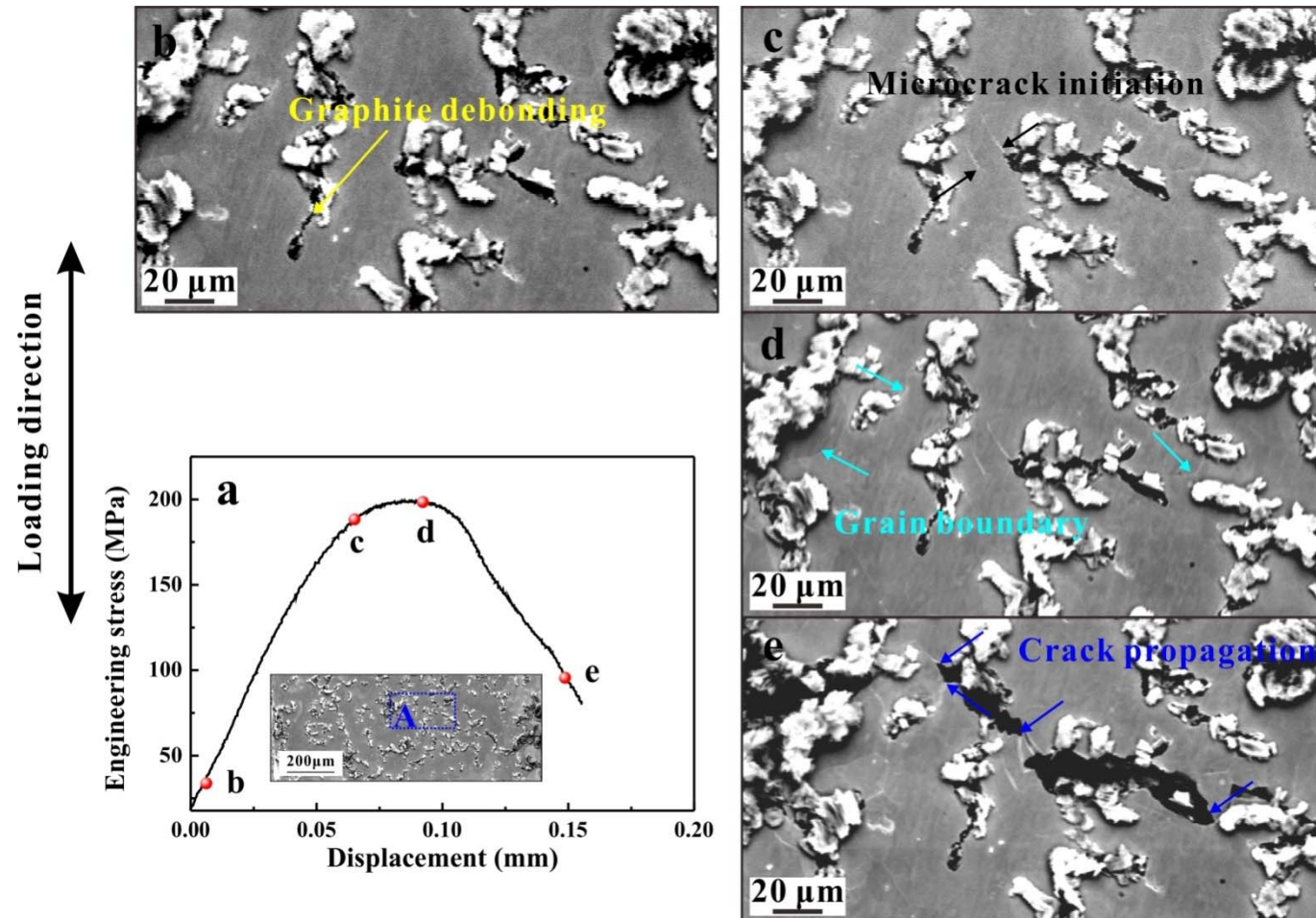
In situ analysis at RT



Graphite crack **Slipping in ferrite**

2. Tensile mechanism of CGI

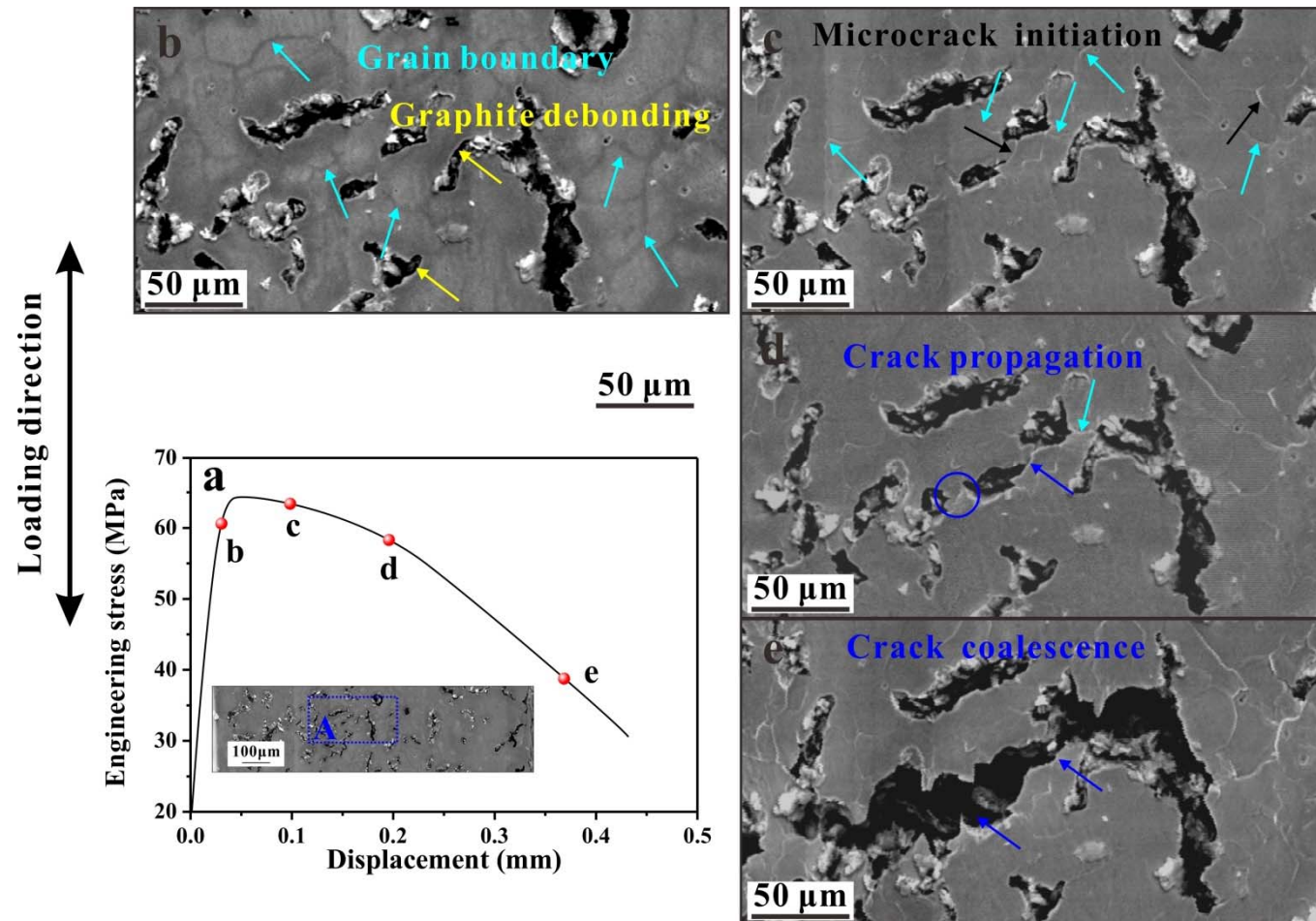
In situ analysis at 450 °C



Inner slipping GB sliding

2. Tensile mechanism of CGI

In situ analysis at 600 °C

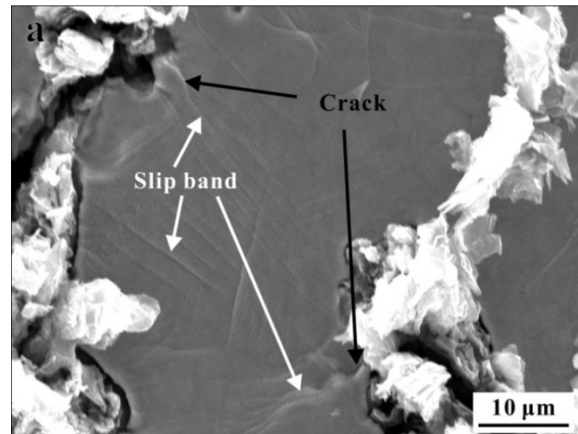


GB sliding in ferrite

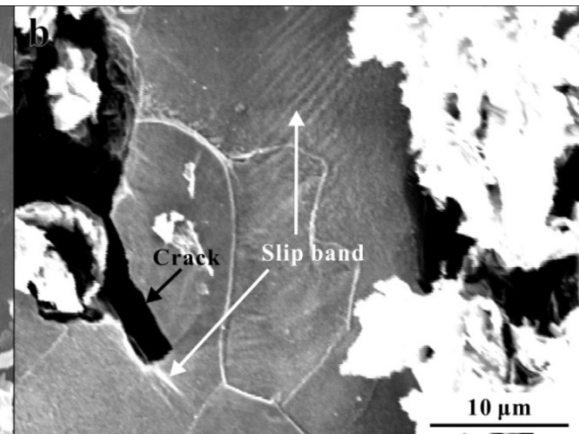
2. Tensile mechanism of CGI

Ferrite analysis after fracture

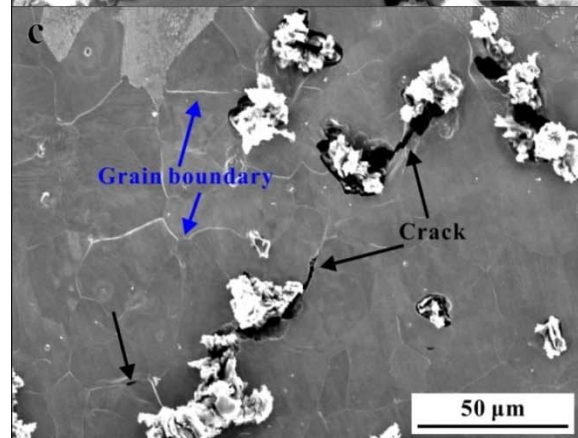
Slipping



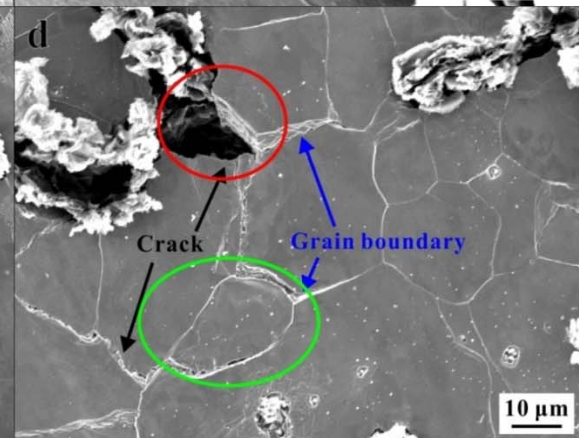
Slipping



Sliding



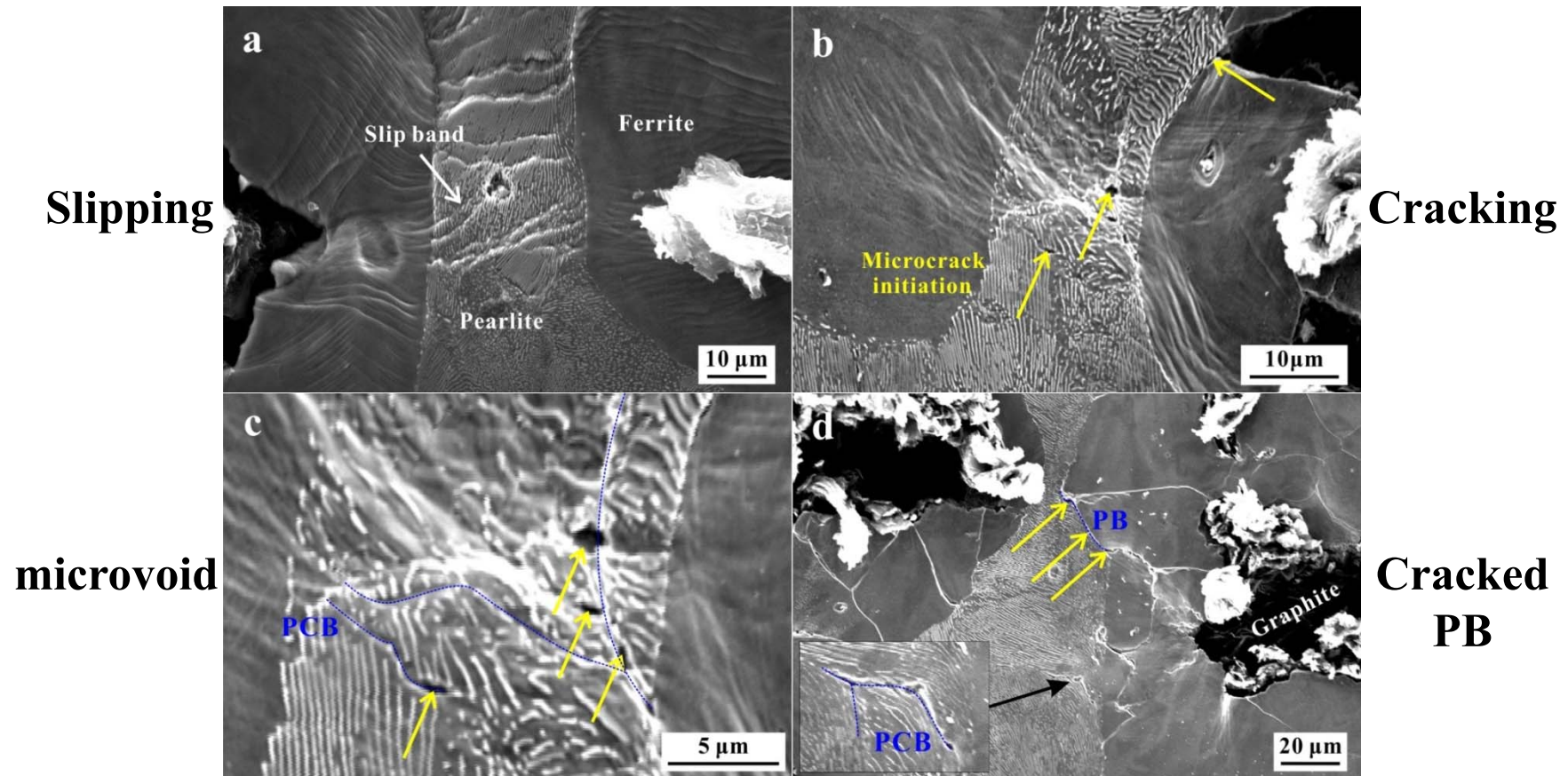
Sliding



(a) 25°C; (b) and (c) 450°C; (d) 550°C

2. Tensile mechanism of CGI

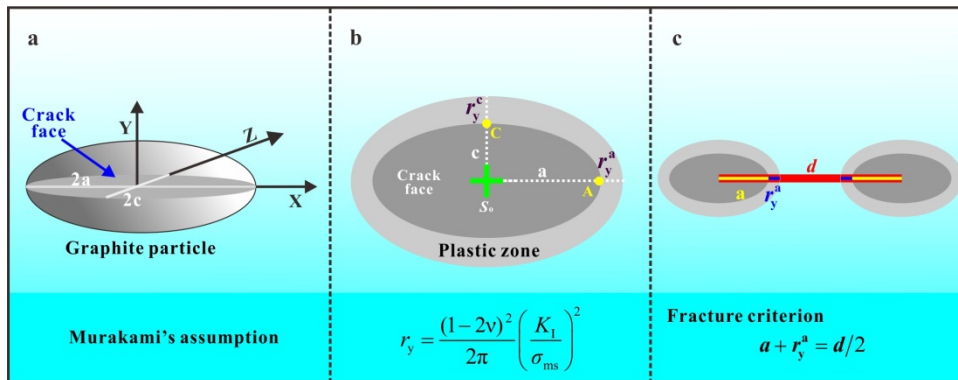
Pearlite analysis after fracture



(a) 25°C; (b) and (c) 450°C; (d) 550°C

3. Tensile quantitative relation

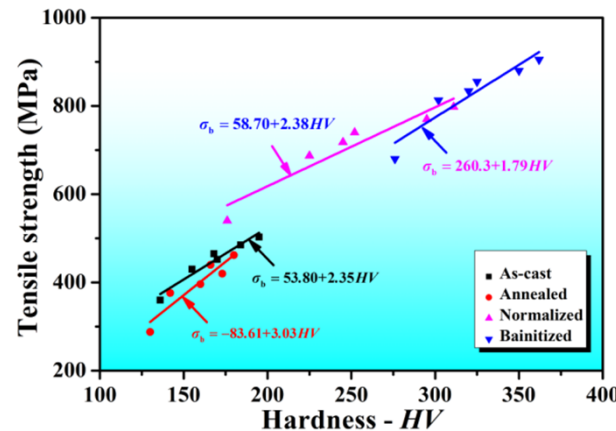
Murakami ' hypothesis



TS of CGI and YS of matrix

$$\sigma_b = \frac{\sqrt{6}\sigma_{ms}}{Y_a} \sqrt{\alpha - 1},$$

$$\alpha = \frac{d}{2a}$$



Takeo Yokobori (T < 450°C)

Dislocation pinning-unpinning

$$\frac{\sigma_{ms}}{\sigma_{ms}^0} = \left(\frac{\dot{\epsilon} E \omega_0}{m k T \sigma_{ms}^0} \right)^{m k T}$$

Strength and temperature:

$$\sigma_b = \sigma_0 \exp(-BT) \cdot \exp(AT \ln T)$$

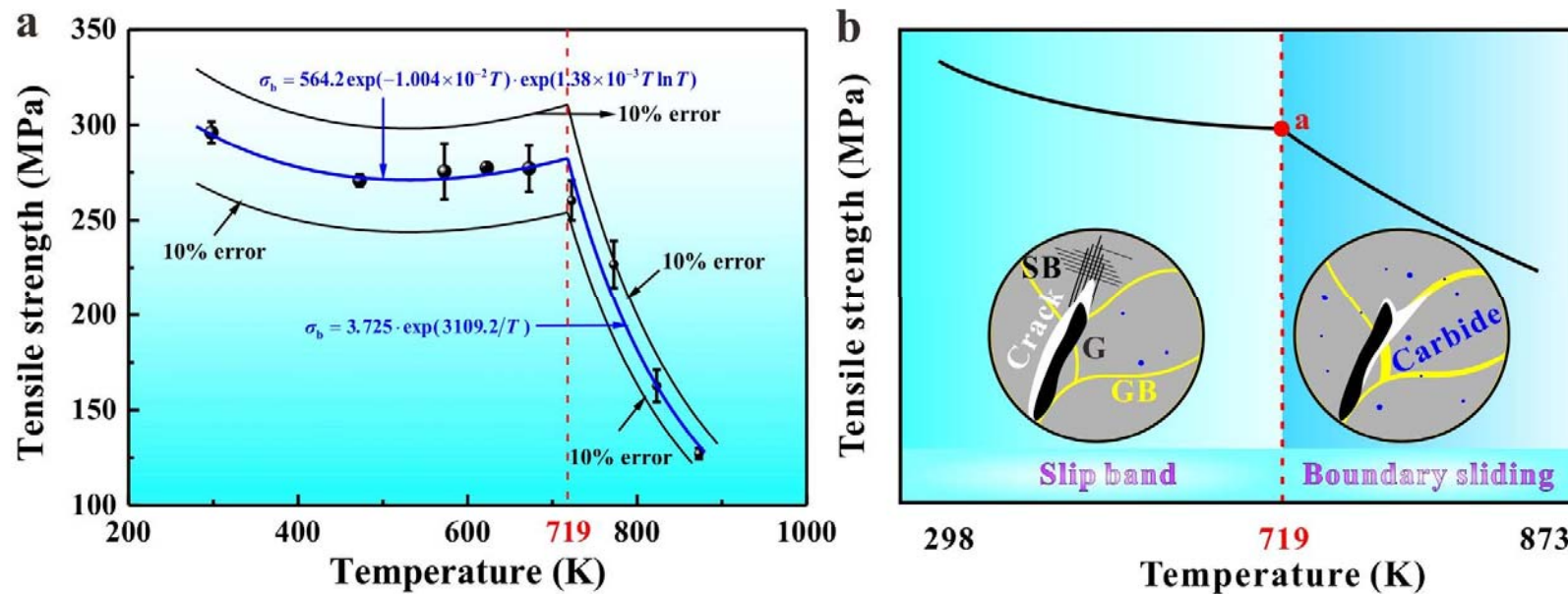
Vacancy diffusion (GB) (T > 450°C)

$$\tau_c = \tau_0 \exp\left(\frac{Q}{RT}\right) \quad \sigma_b = \frac{\tau_c}{\cos(\bar{\theta}) \sin(\bar{\theta})} \left(1 - \frac{\Delta l_0}{l_0}\right)$$

Strength and temperature:

$$\sigma_b = \sigma_0 \exp\left(\frac{Q}{RT}\right)$$

4. Brief summary



Tensile strength (a) and mechanism (b) evolutions of CGI with temperature

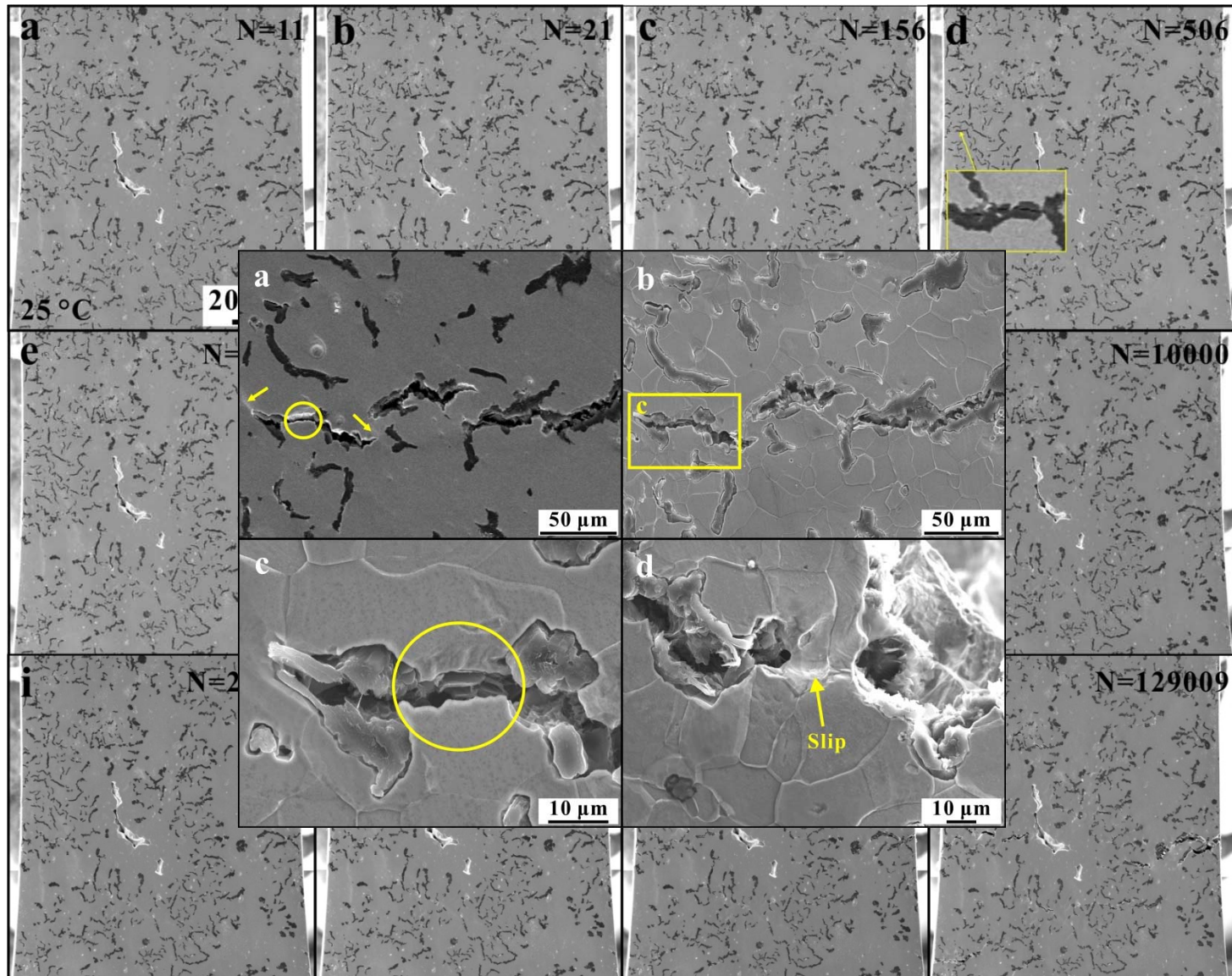
三、高温疲劳损伤机制与强度预测

Fatigue damage mechanism and strength prediction

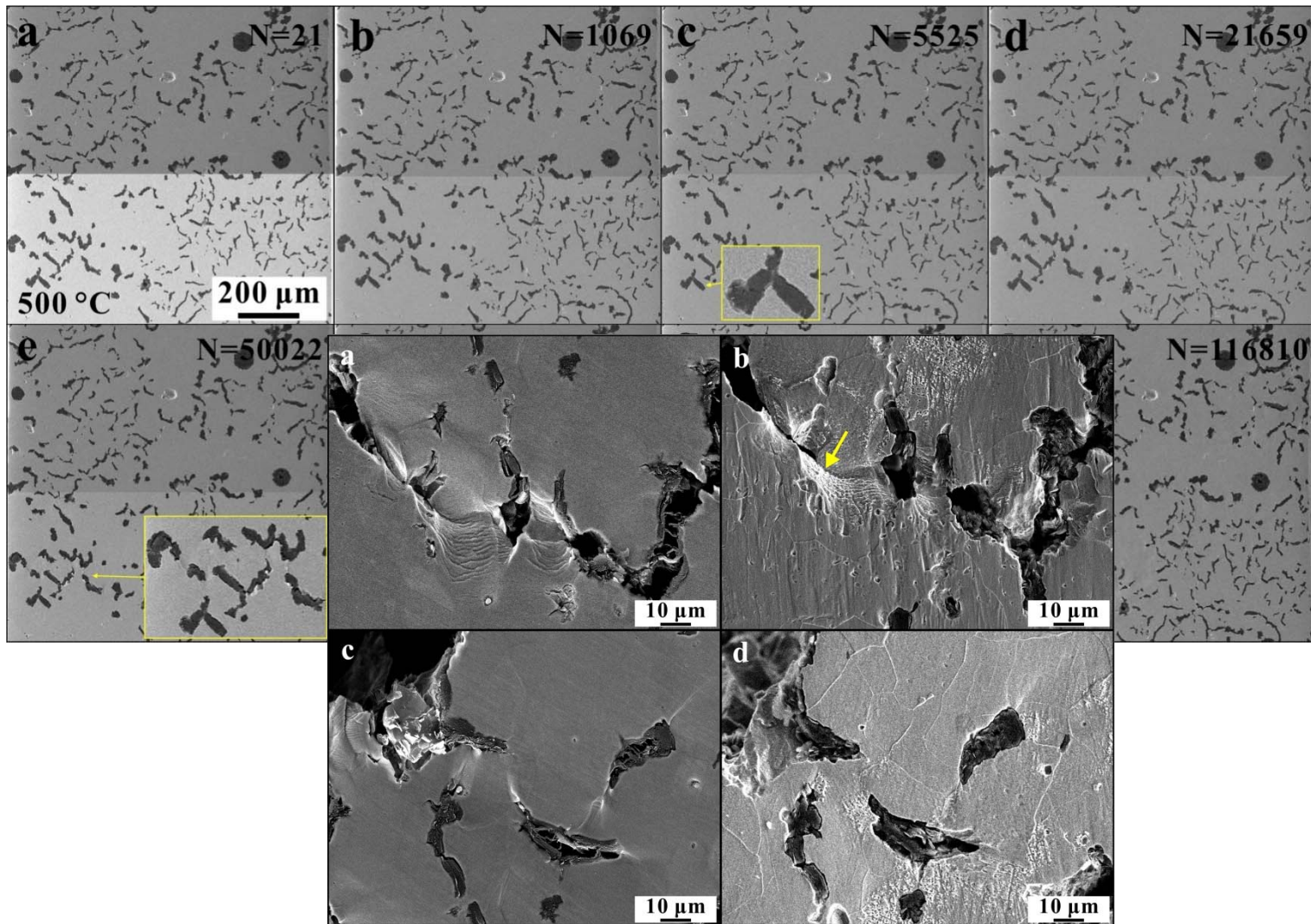
1. Mechanism analysis of CGI

S. I

25°C

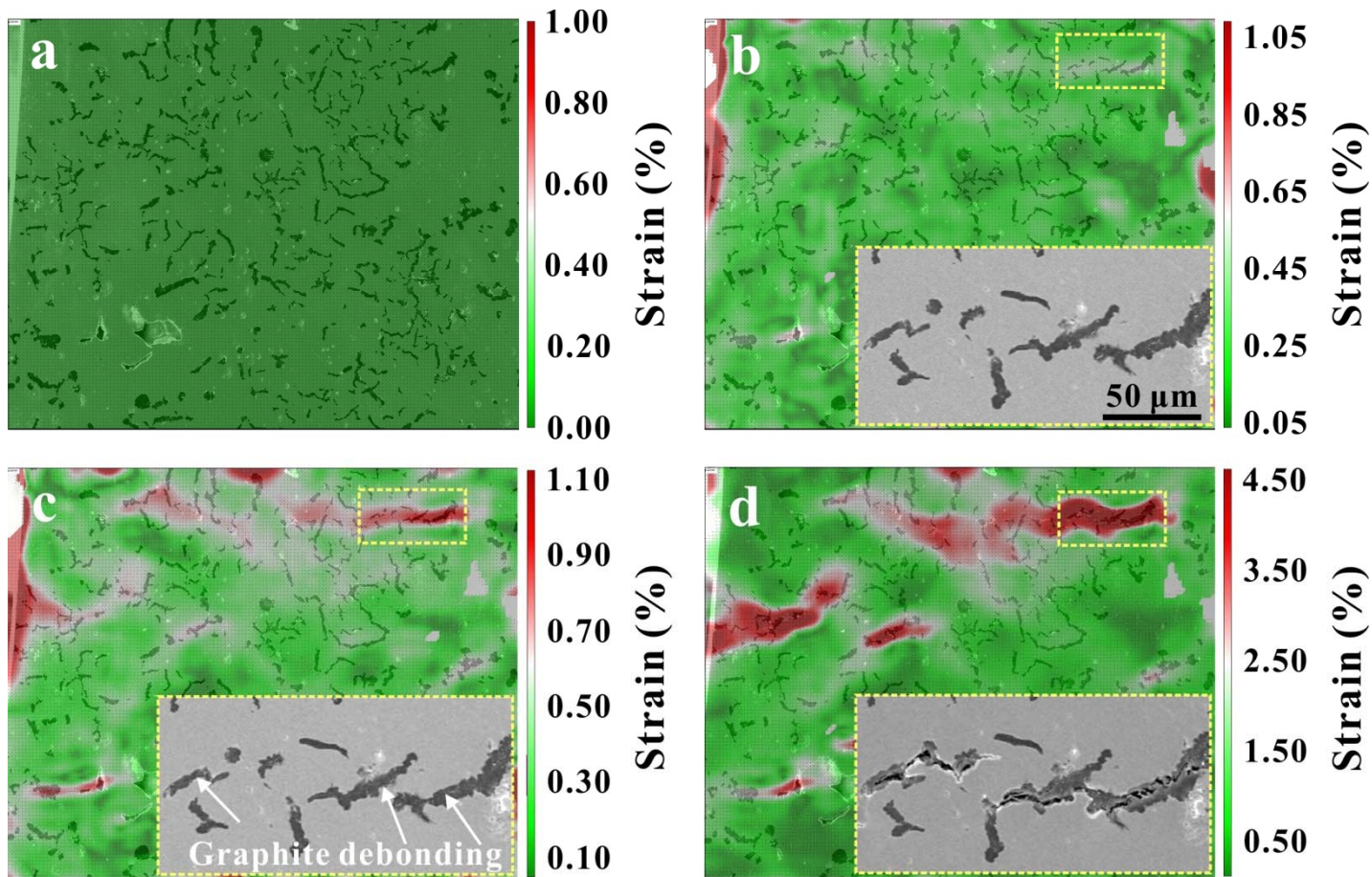


1. Mechanism analysis of CGI



500 °C

1. Mechanism analysis of CGI

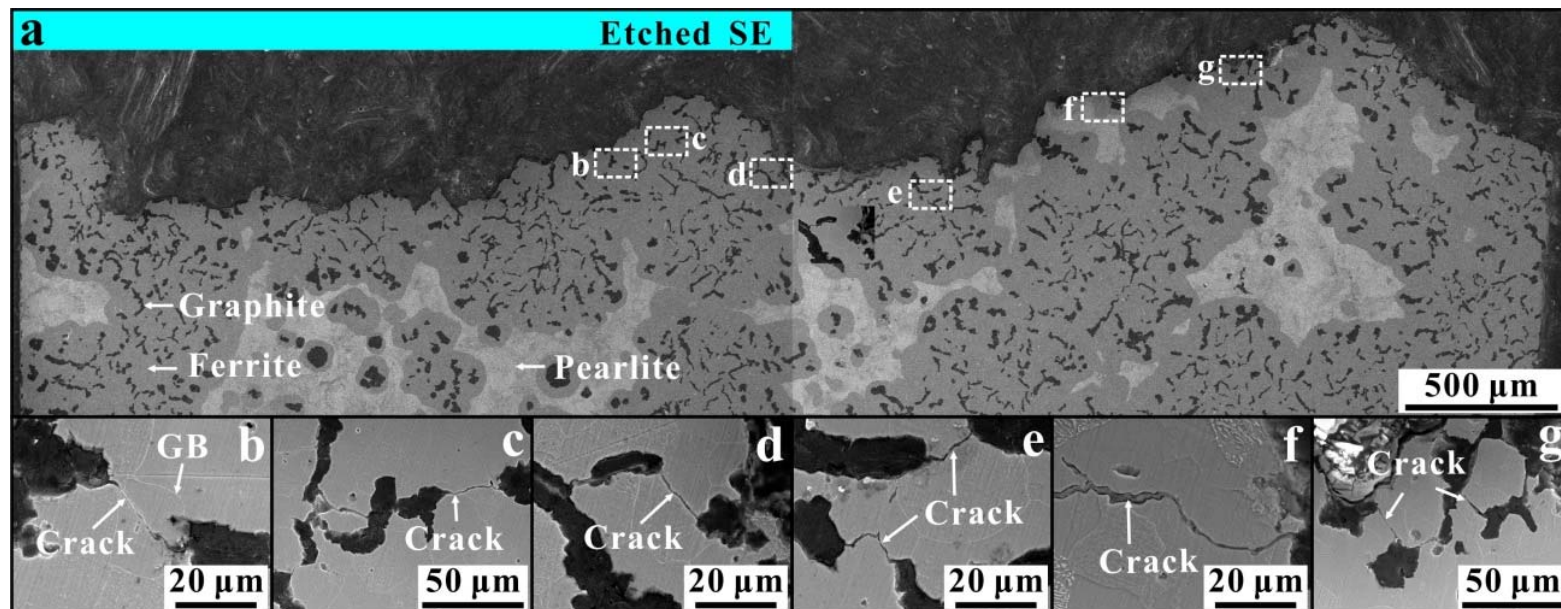


25°C S. I : (a) N= 0; (b) N=156; (c) N=506 and (d) N=120009

1. Mechanism analysis of CGI

25°C

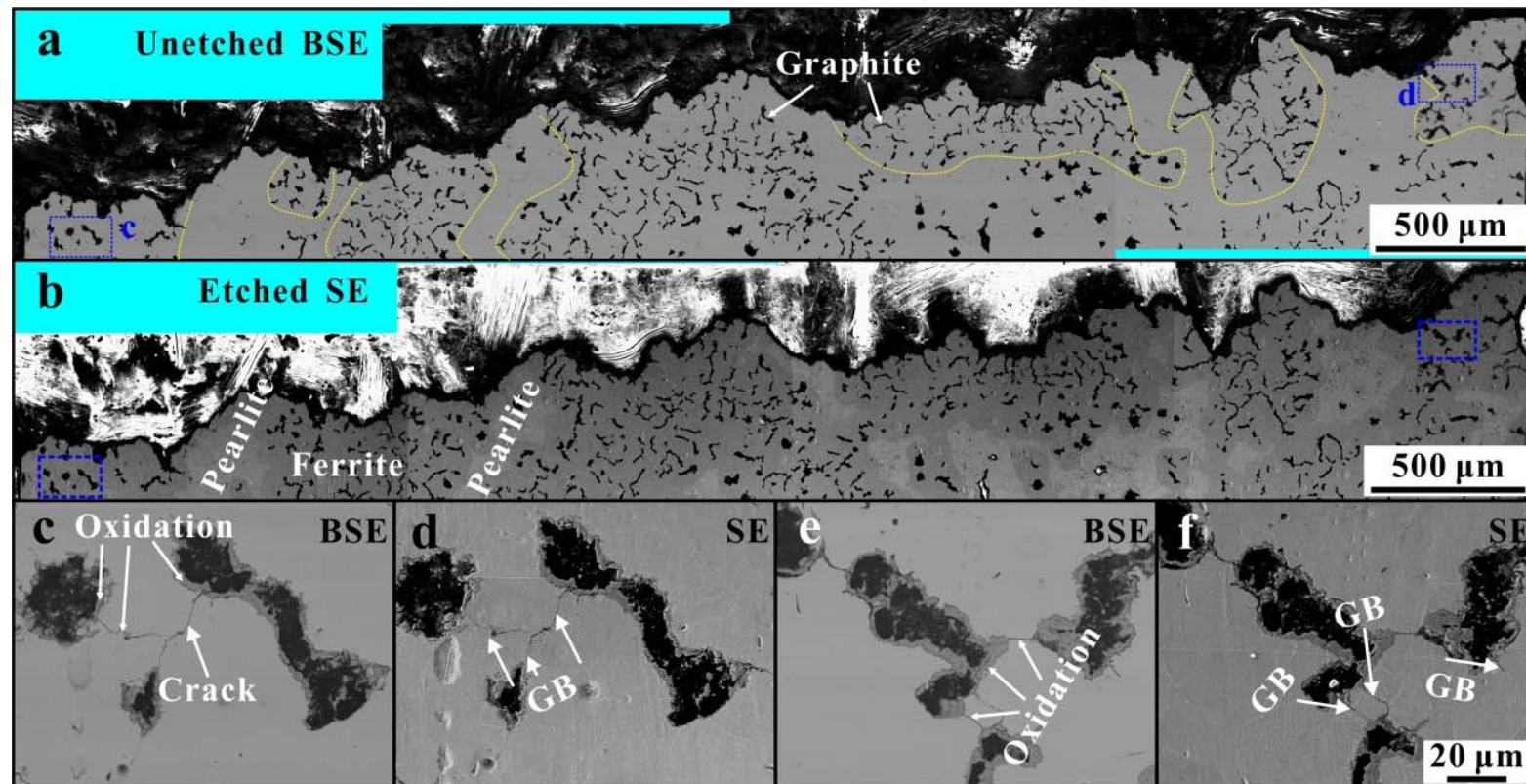
S. I



Typical profile morphology : $\sigma_a = 160\text{MPa}$, $N_f = 489151$

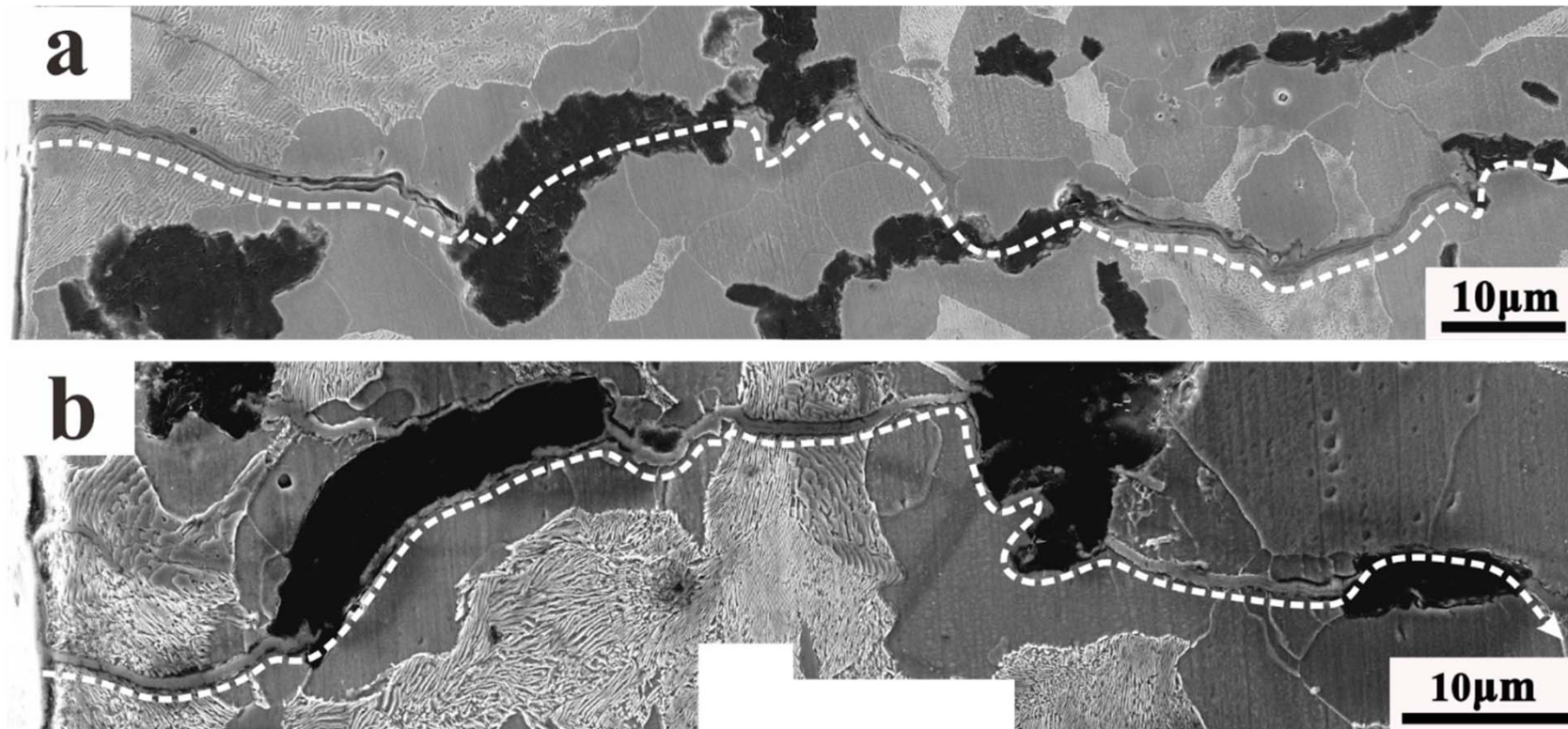
2. Mechanism analysis of CGI

500°C S. I



Typical profile morphology : $\sigma_a = 170\text{MPa}$, $N_f = 163812$

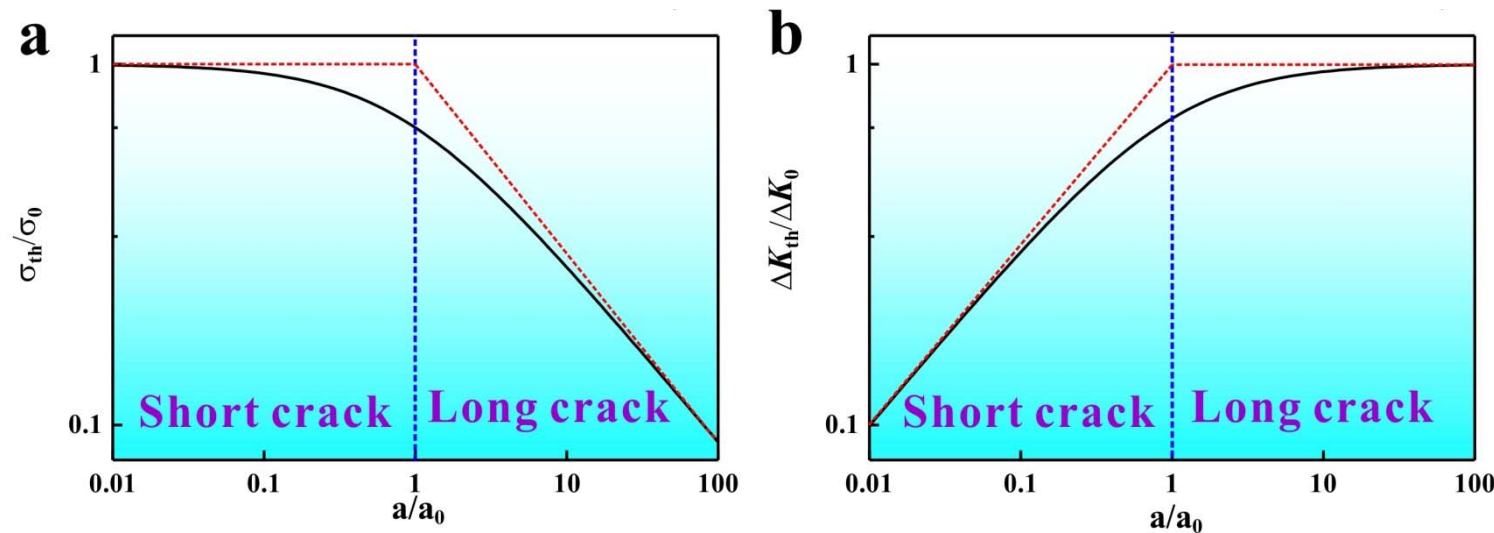
1. Mechanism analysis of CGI



S.III: (a) 400 °C; (b) 500 °C

Fatigue crack initiation and propagation

2. Quantitative relation of CGI



Critical crack size

BSB model

Slip band blocked by grain boundary

$$a_0 = \frac{1}{\pi} \left(\frac{\Delta K_{th}}{\Delta \sigma_e} \right)^2,$$

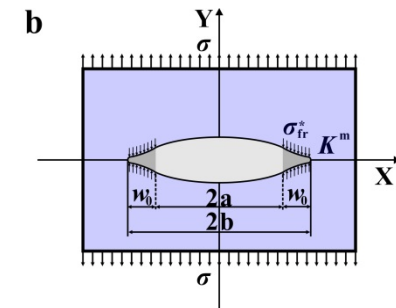
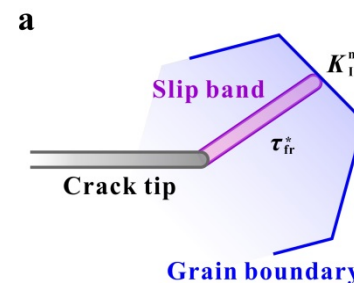
$$\sigma_w = \frac{\sigma_e}{\sqrt{a/a_0 + 1}}$$

$$a = a_0$$

a_0 different

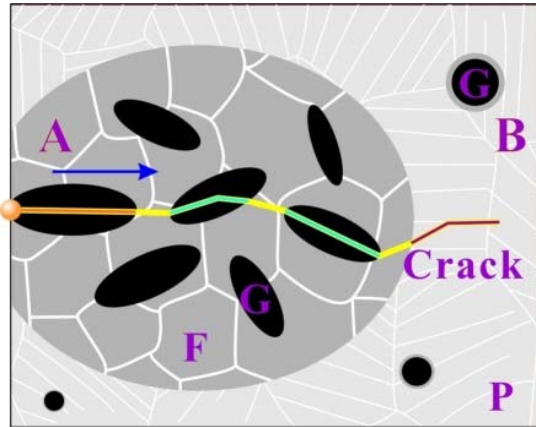
a - Semi crack length

a_0 -Plastic zone size



Kitagawa H, et al. ASM, Metals Park, Ohio. 1976.El-Haddad MH, et al. *Eng. Fract. Mech.* 1979 11: 573.Tanaka K, et al. *Int. J. Fracture.* 1981 17: 519.

2. Quantitative relation of CGI



Graphite

$$\sqrt{Area_g}$$

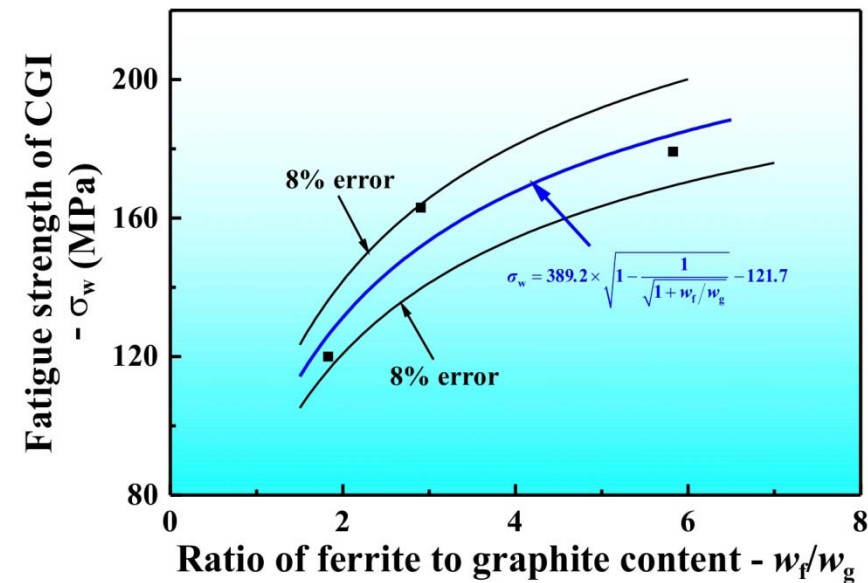
Ferrite

$$\sqrt{Area_g + Area_f} - \sqrt{Area_g}$$

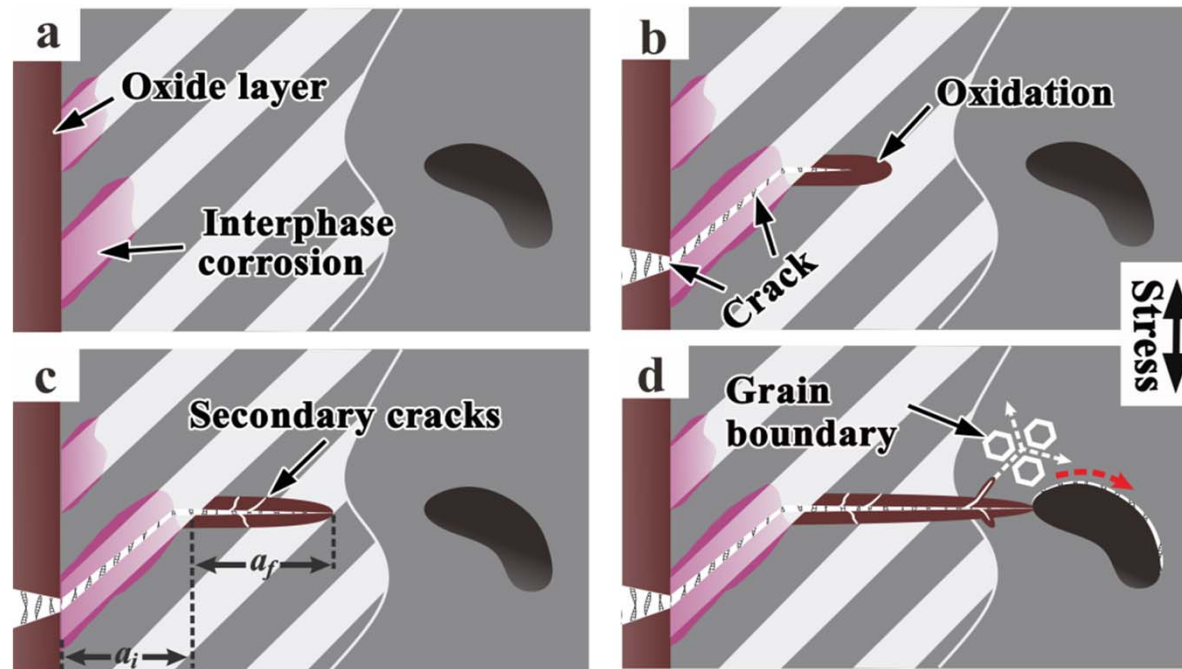
Fatigue strength of CGI

$$\sigma_w = \sigma_w^f \sqrt{1 - \frac{1}{\sqrt{1 + Area_f / Area_g}}}$$

$$\sigma_w = \sigma_w^f \sqrt{1 - \frac{1}{\sqrt{1 + w_f / w_g}}}$$

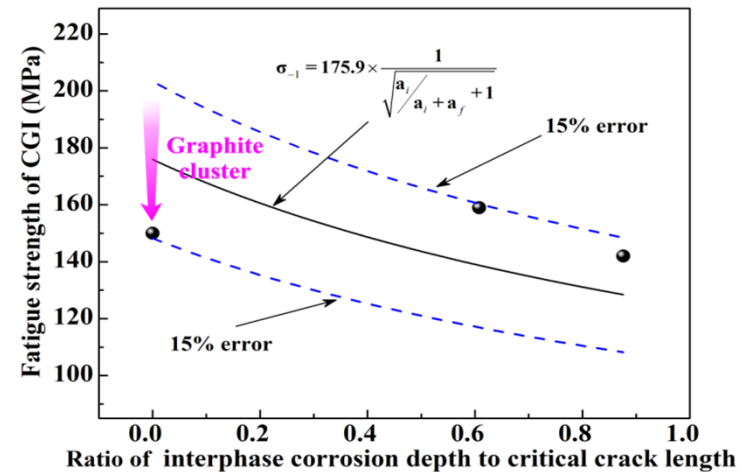


2. Quantitative relation of CGI



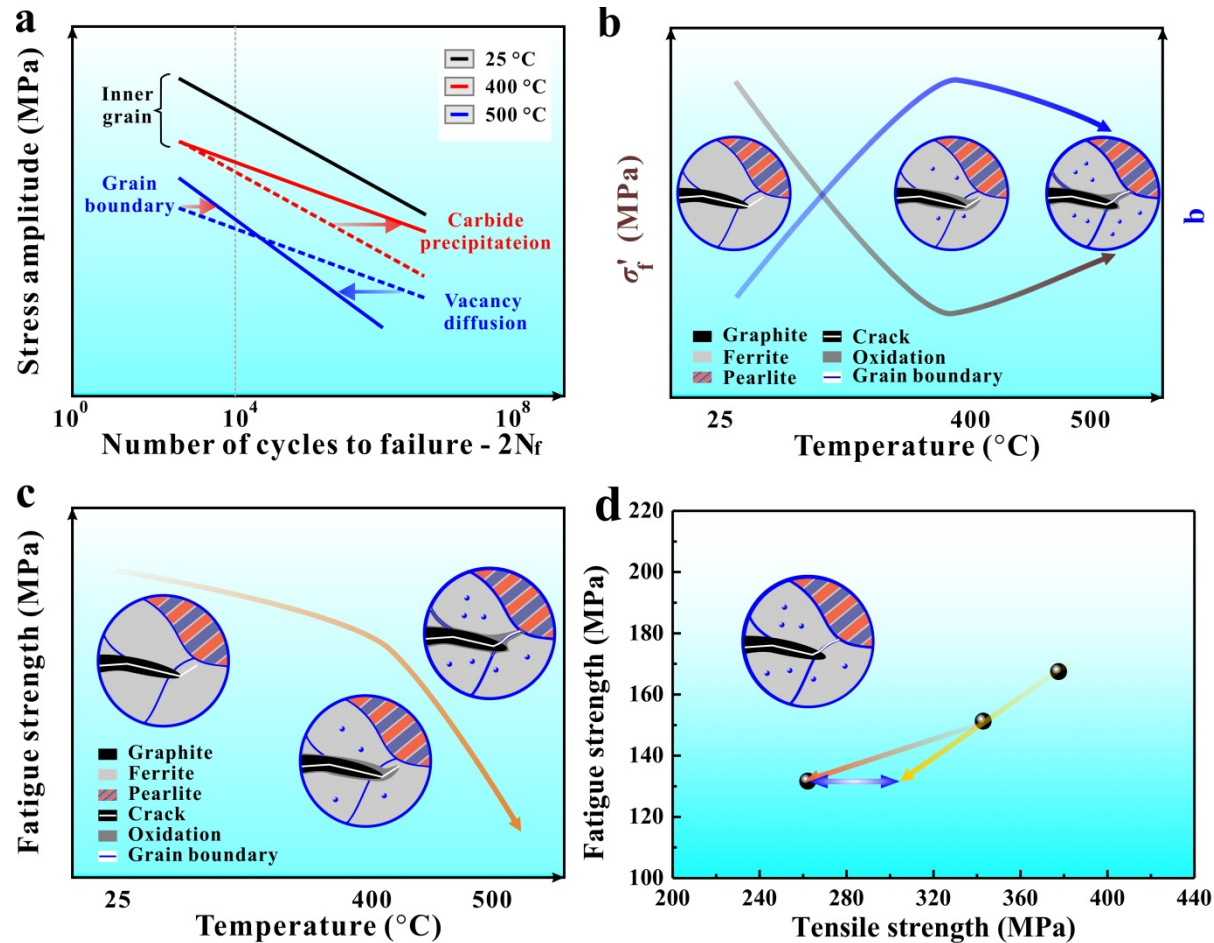
Fatigue strength of CGI

$$\sigma_w = \sigma_w^f \frac{1}{\sqrt{1 + \frac{a_i}{a_i + a_f}}}$$



3. Brief summary

Temperature influence

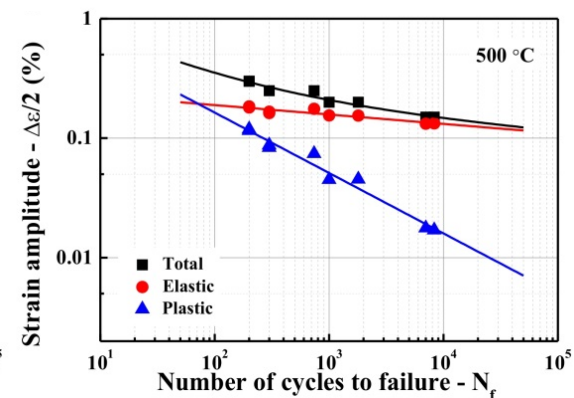
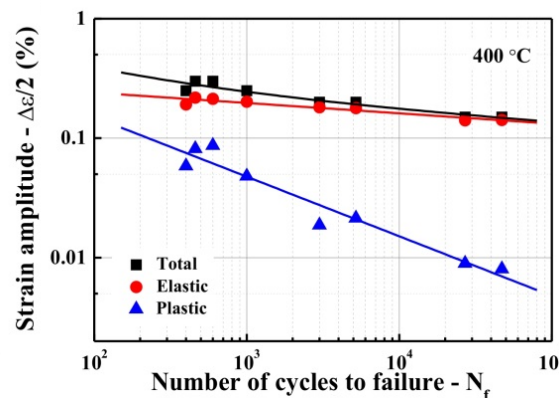
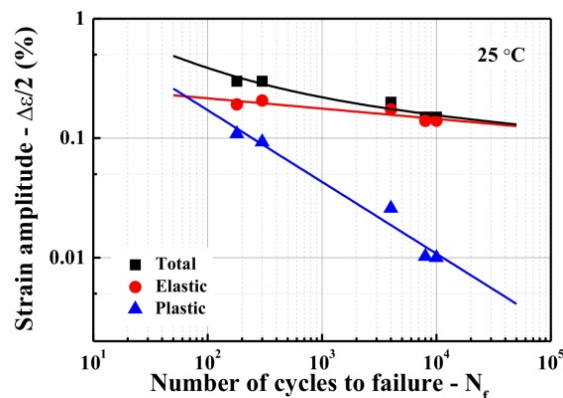
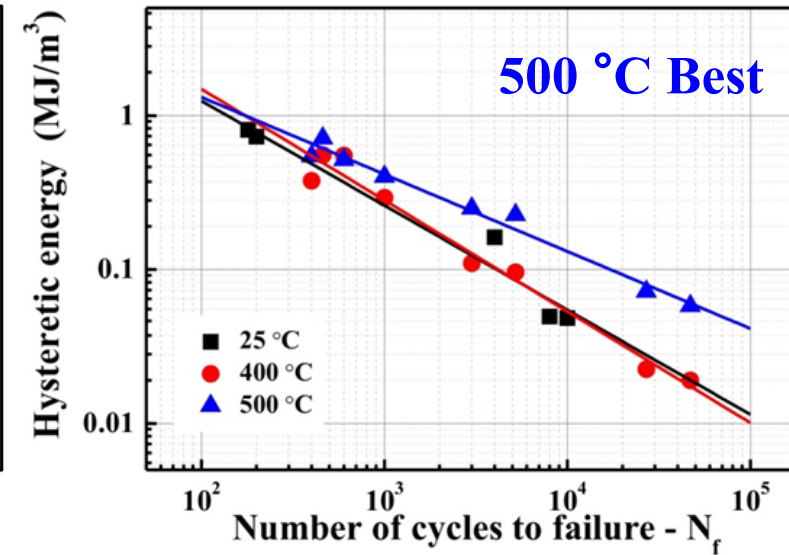
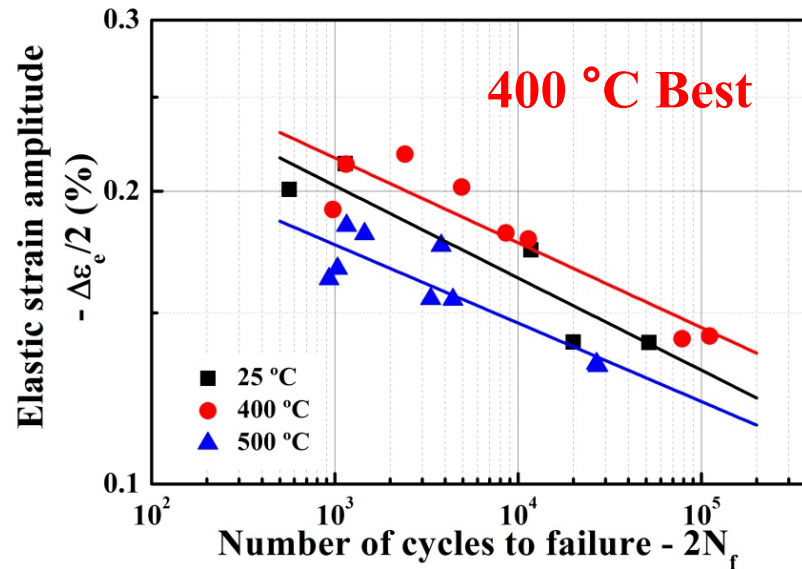


(a) S-N ; (b) Basquin relation;(c) fatigue strength; (d) relation between tensile and fatigue strengths

四、疲劳寿命预测模型与损伤机制

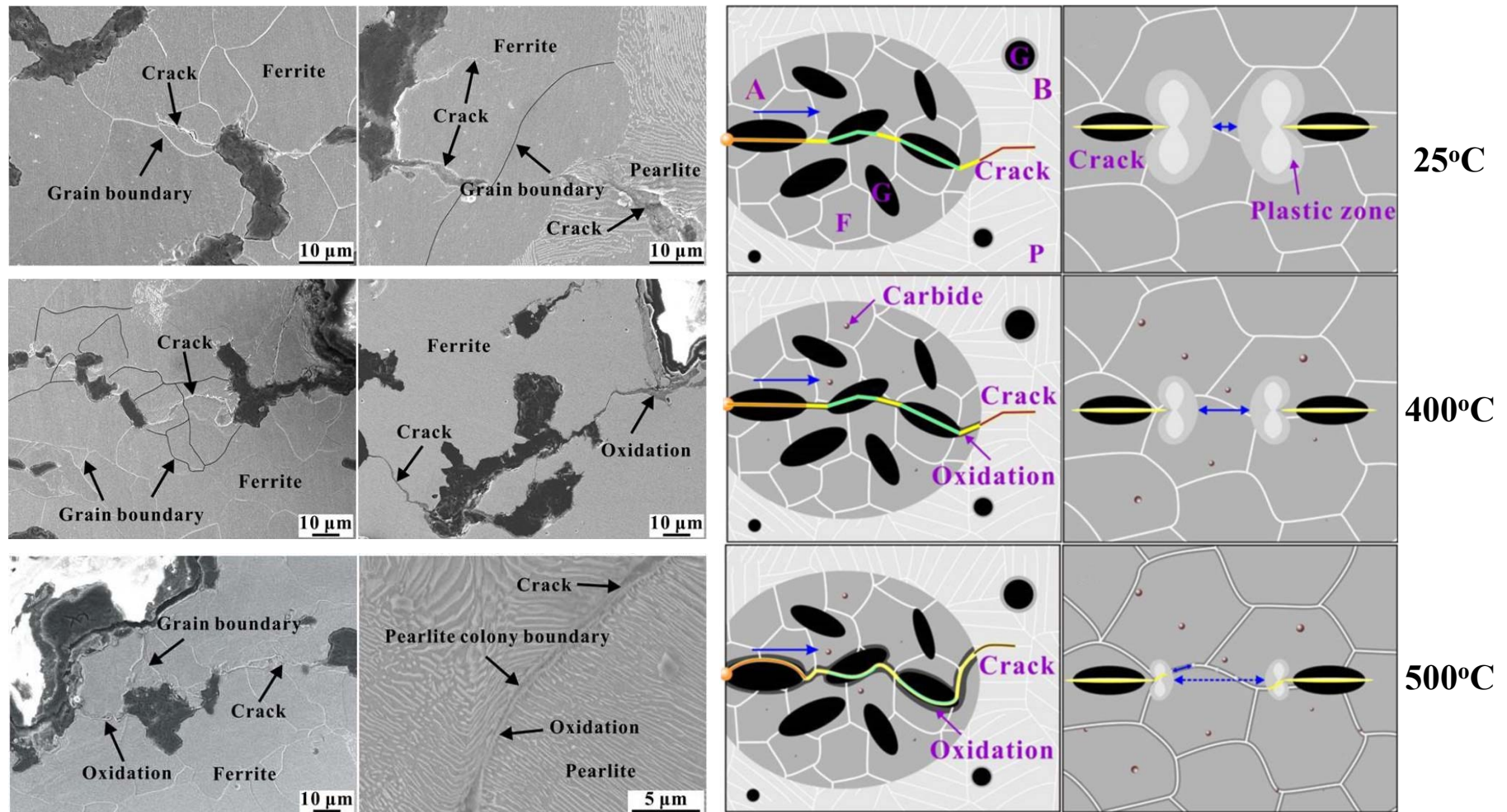
Fatigue life prediction model and damage mechanism

1. LCF properties and mechanism at HTs



Non-monotonic change of fatigue life with temperature

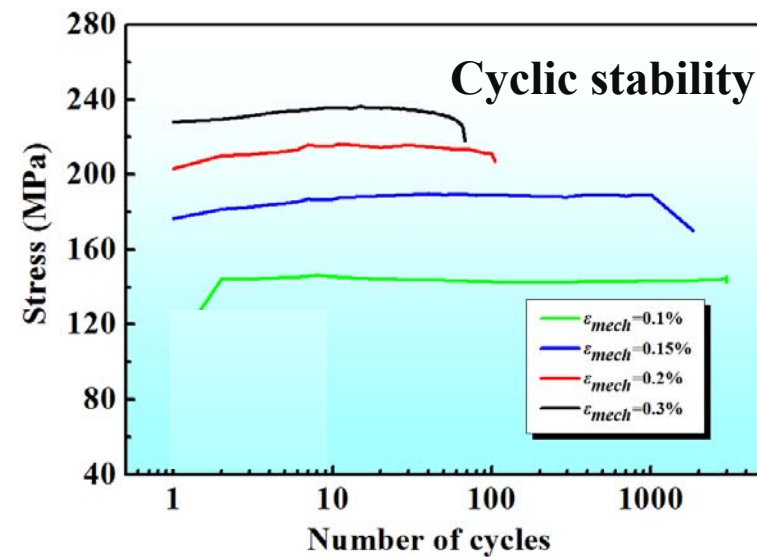
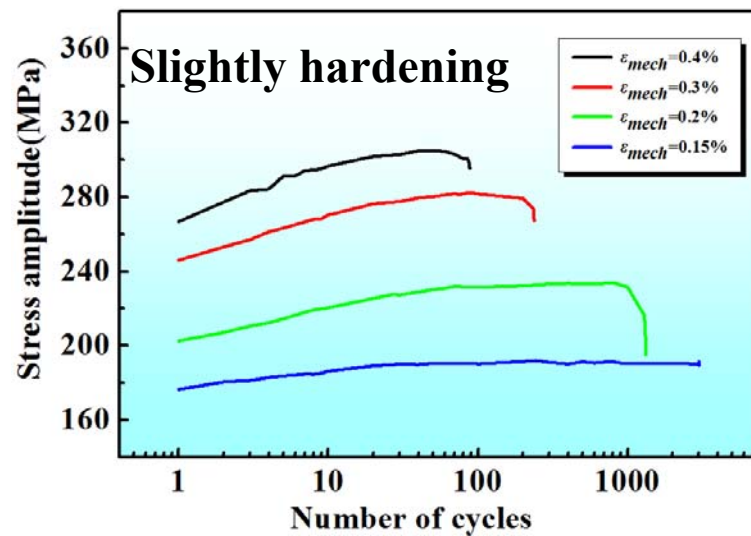
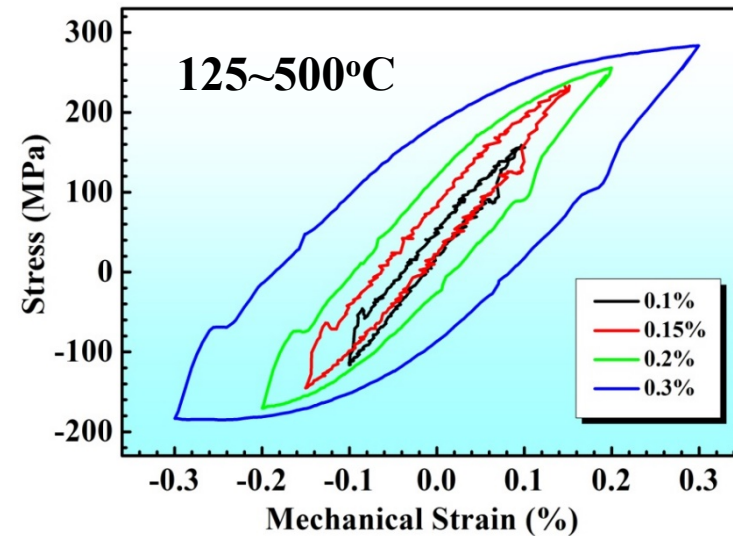
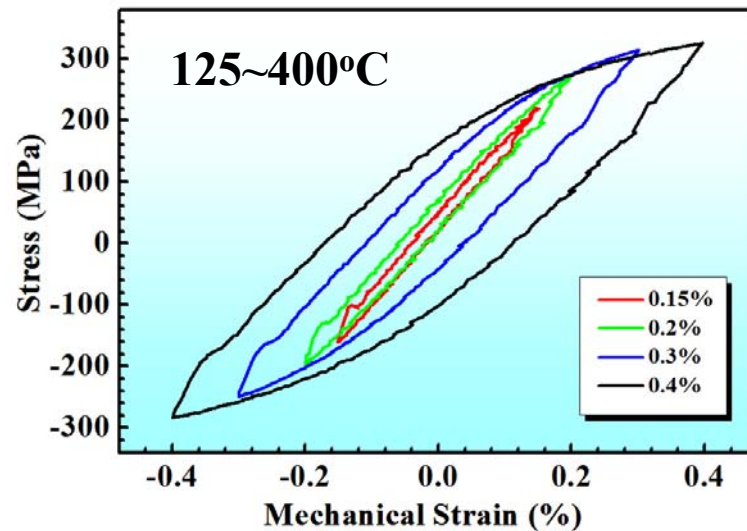
1. LCF properties and mechanism at HTs



Crack propagation

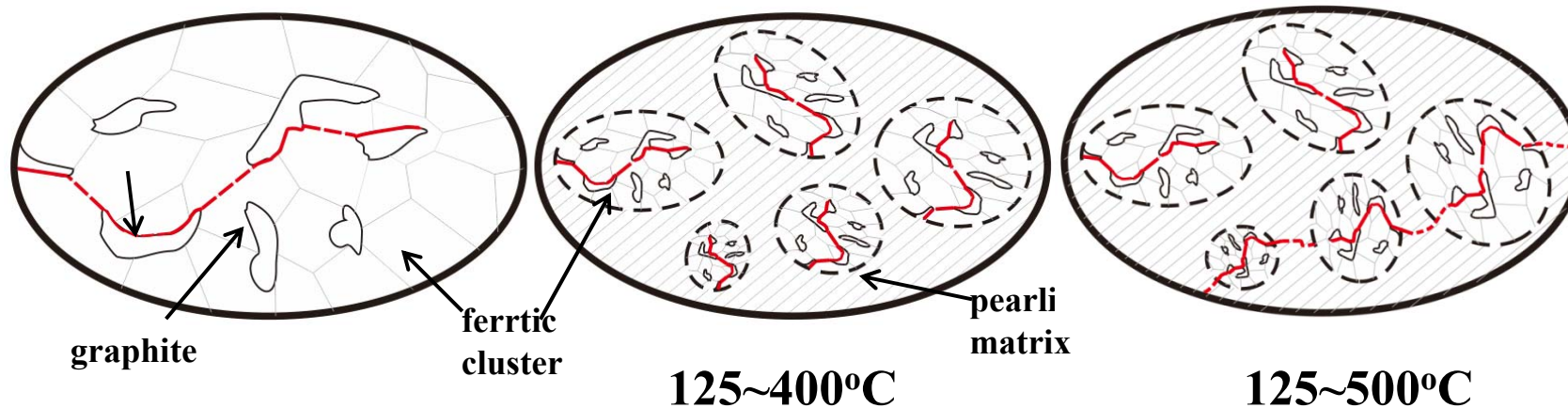
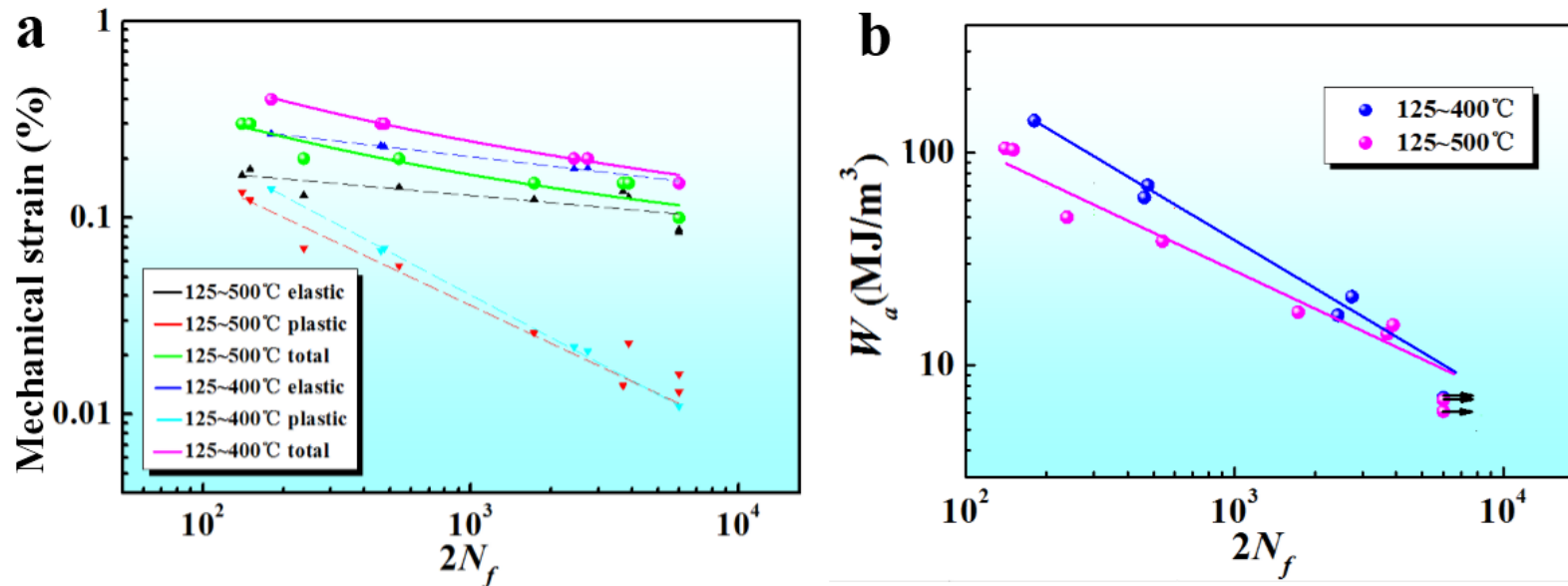
Damage mechanism

2. TMF properties and mechanism



Hysteresis loop and cyclic behavior curves

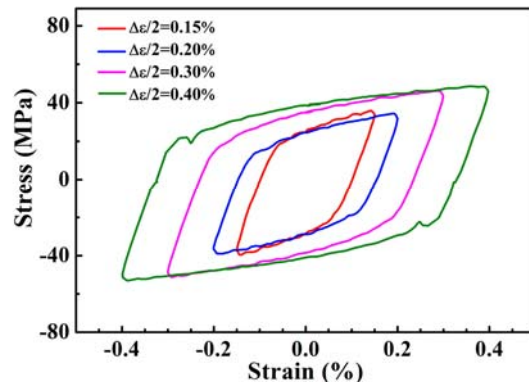
2. TMF properties and mechanism



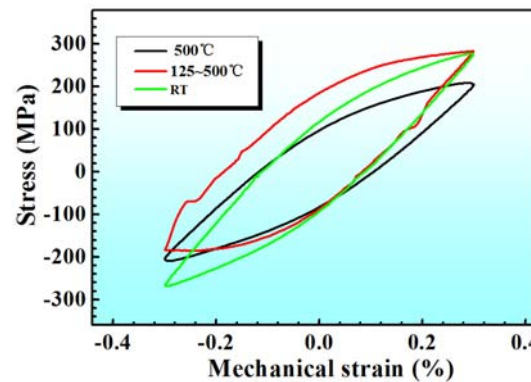
Behavior&mechanism: LCF at HTs $\approx$ TMF No creep & oxidation

3. Simply quantitative relation: Hysteresis energy

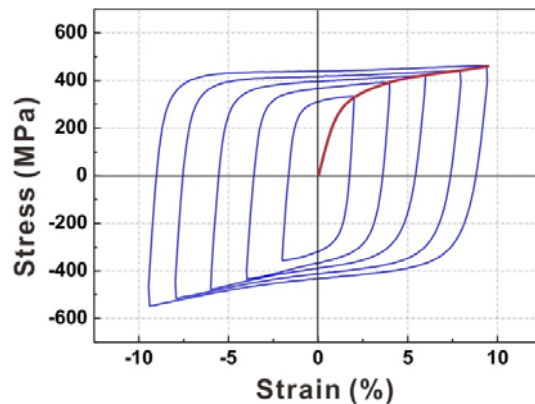
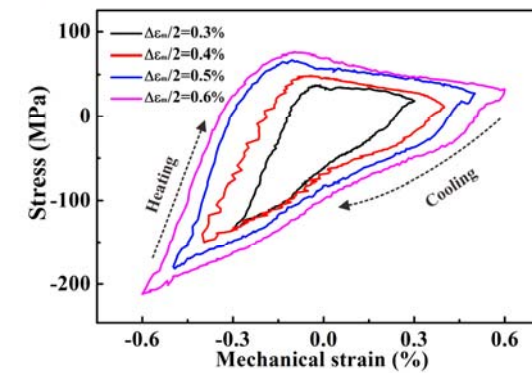
LCF at HTs



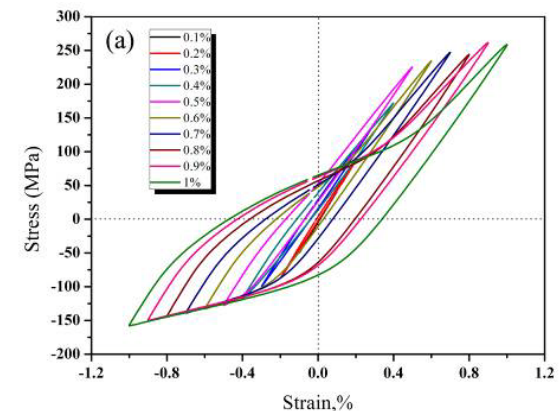
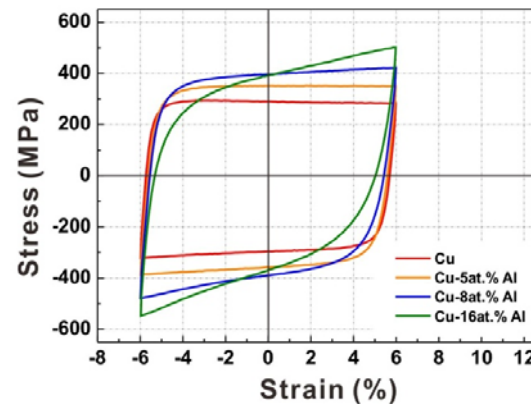
Bridge: Energy



TMF

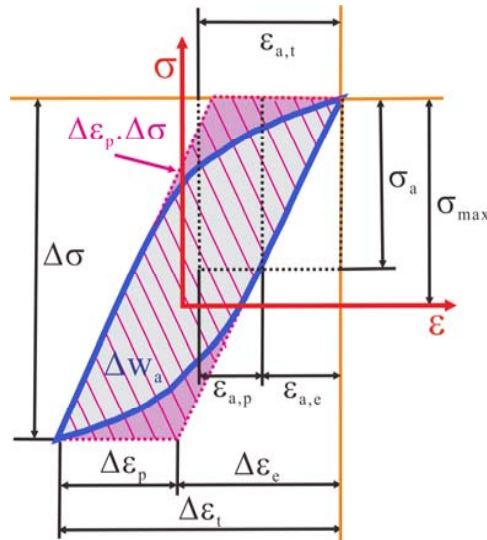


LCF at RT

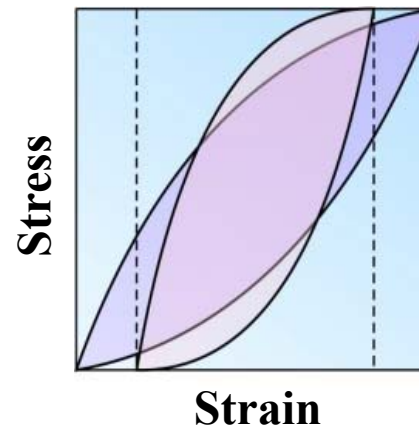


Different mechanism, similar form,
reflected by **hysteresis energy**

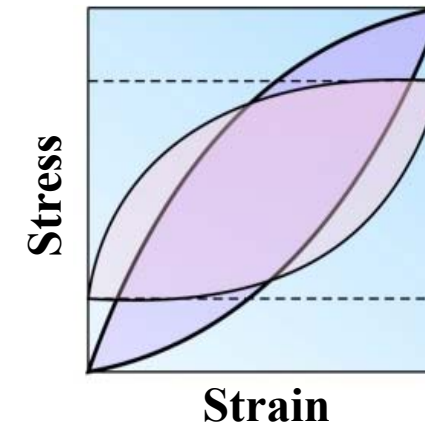
3. Simply quantitative relation: Hysteresis energy



Basquin relation



Coffin-Manson relation



Prediction model at RT : Hysteresis energy reflect damage

**Energy
model**

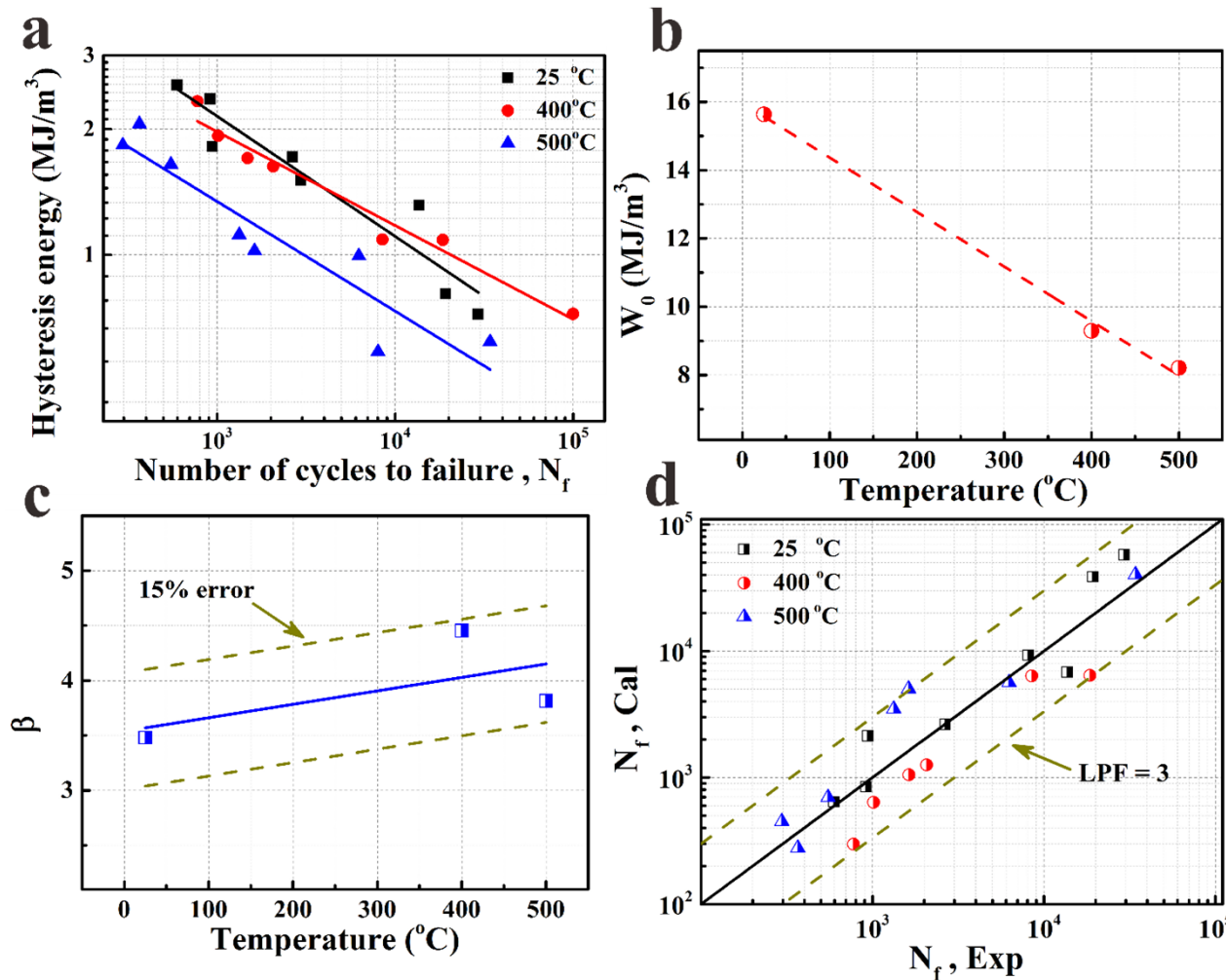
$$D_i = \left(\frac{W_i}{W_0} \right)^\beta$$

$$\text{Damage} = \left(\frac{\text{Energy}}{\text{Toughness}} \right)^\beta$$

**External
Internal**

Energy model can obtain life prediction and optimization at RT

3. Simply quantitative relation: Hysteresis energy



$$N_f = (W_0 / W_s)^{1/\beta}$$

$$W_0 = aT + b$$

$$1/\beta = mT + n$$

$$N_f = \left[\frac{(aT + b)}{W_s} \right]^{1/mT + n}$$

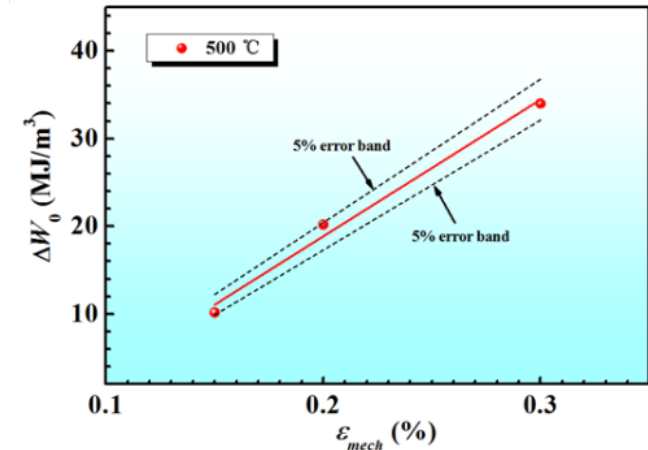
Introduce the **temperature** parameter, obtain LCF model at **HTs**

3. Simply quantitative relation: Hysteresis energy for TMF

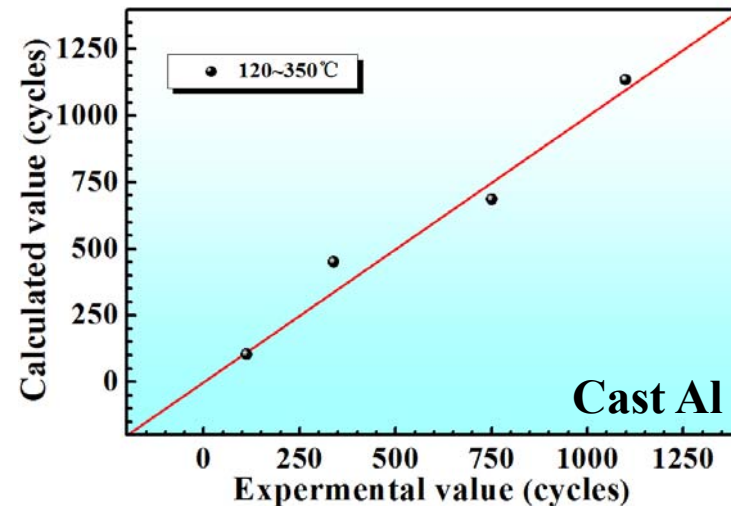
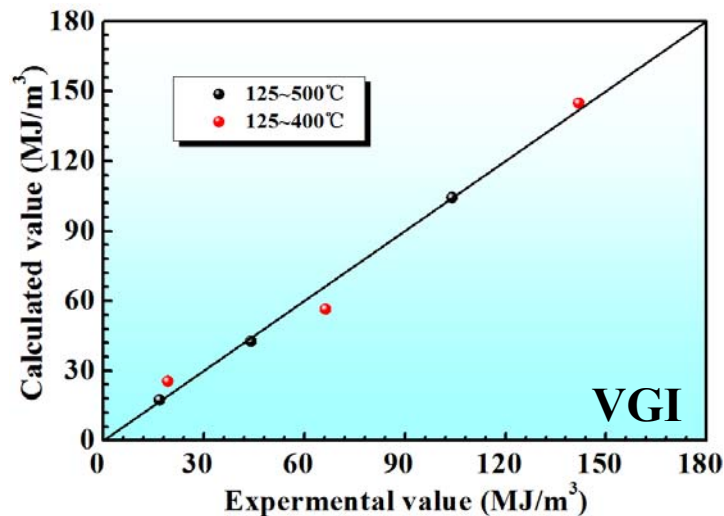
TMF energy

LCF at constant T

$$W_{a, TMF} = A + K \varepsilon_{mech} + W_{a, LCF}$$



Application: replace **TMF** by LCF at constant temperature

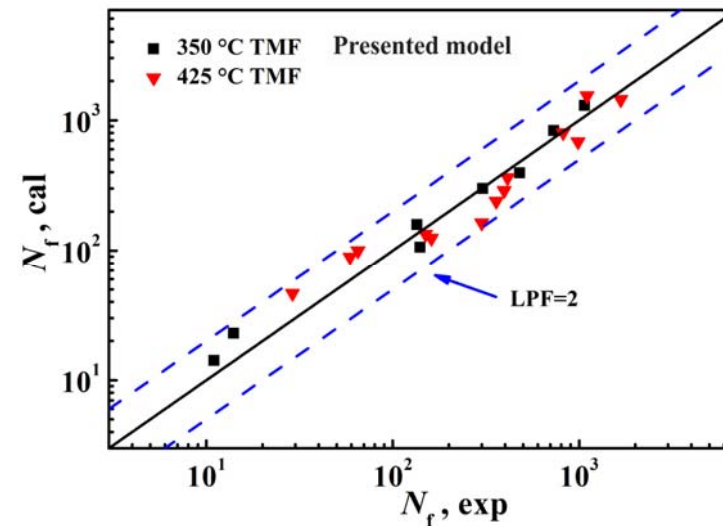
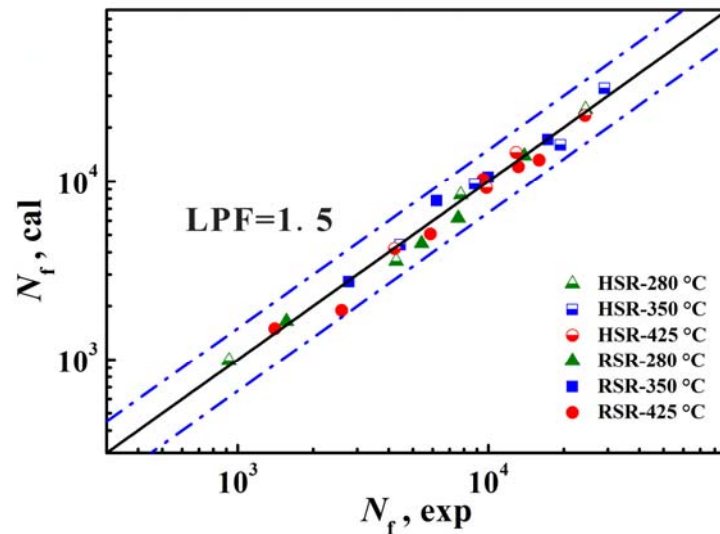
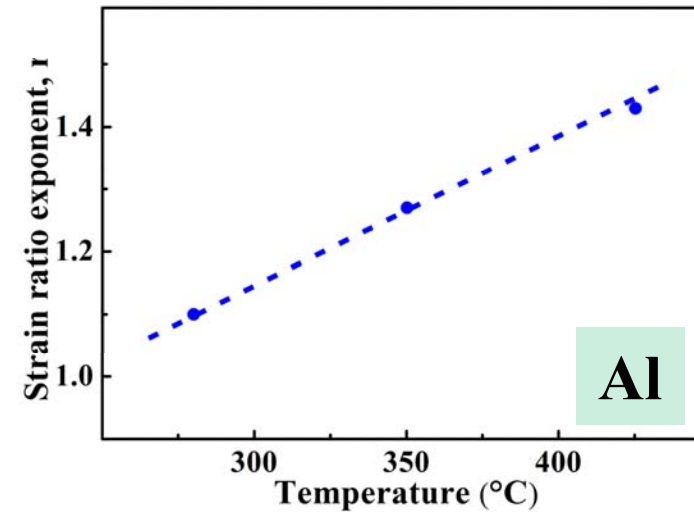


Simplify test, reduce cost, increase efficiency

4. Simply quantitative relation: Hysteresis energy for Al-Si

$$N_f = \dot{\epsilon}^{1-r} \cdot \left(\frac{W_0}{W_s} \right)^{1/\beta}$$

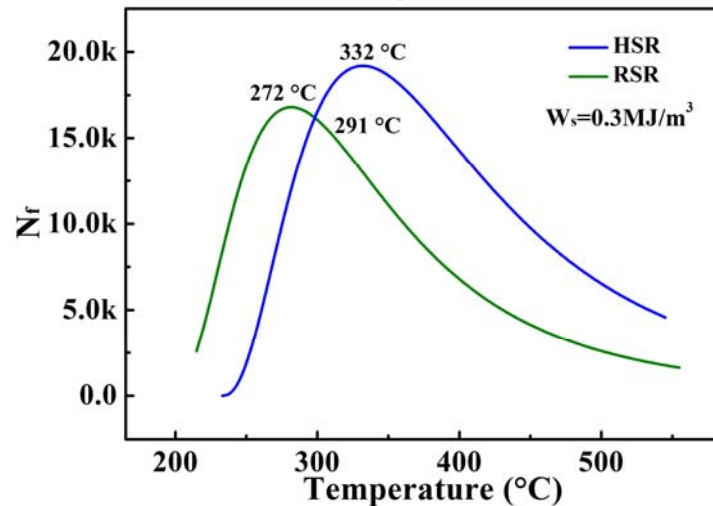
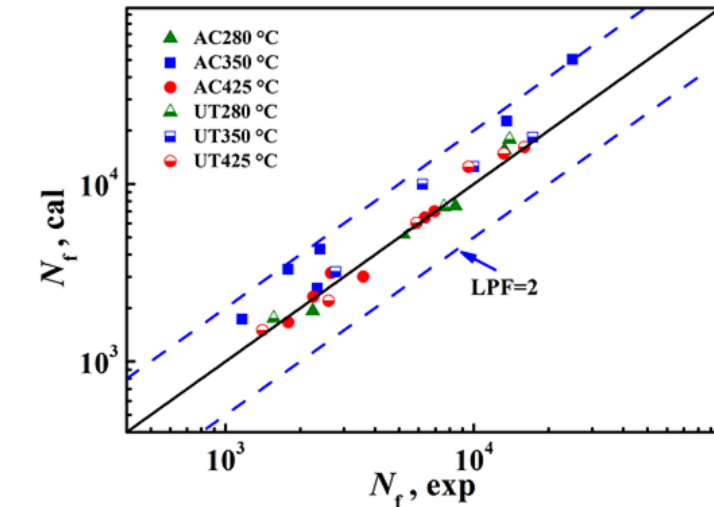
$$W_s^{TMF} = W_0 \left(\frac{\nu_{LCF}}{\nu_{TMF}} \right)^{-\beta(1-k)} \cdot N_f^{-\frac{1}{\beta}}$$



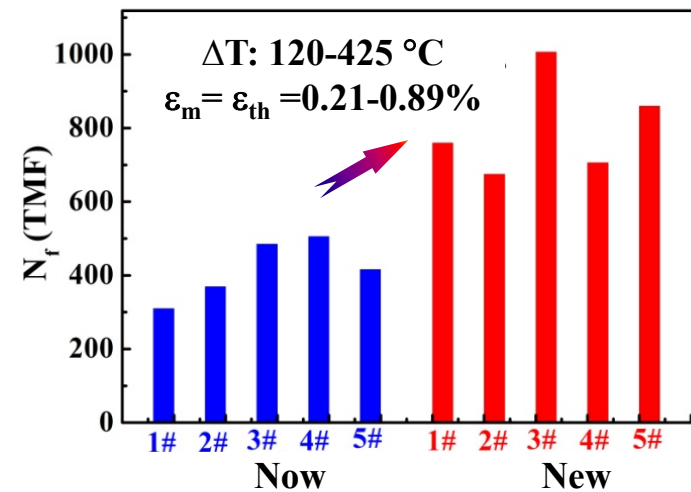
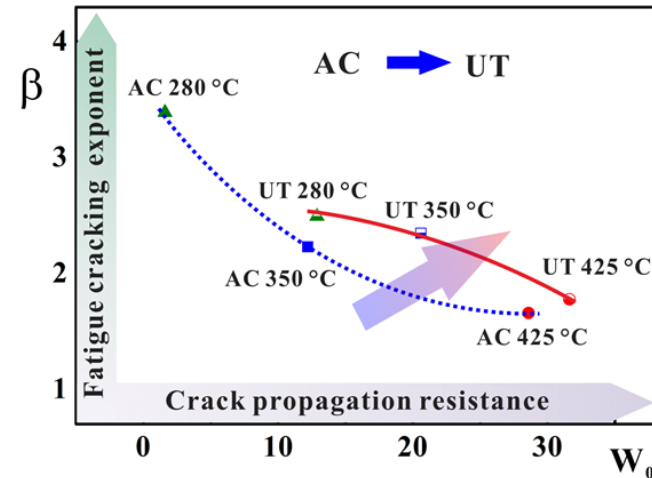
Strain ratio r : Introduction of r can reflect coupling damage mechanism of creep and fatigue

4. Simply quantitative relation: Hysteresis energy for Al-Si

Life prediction



Property optimization

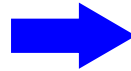


New model **predict well, optimize** property

5. Brief summary



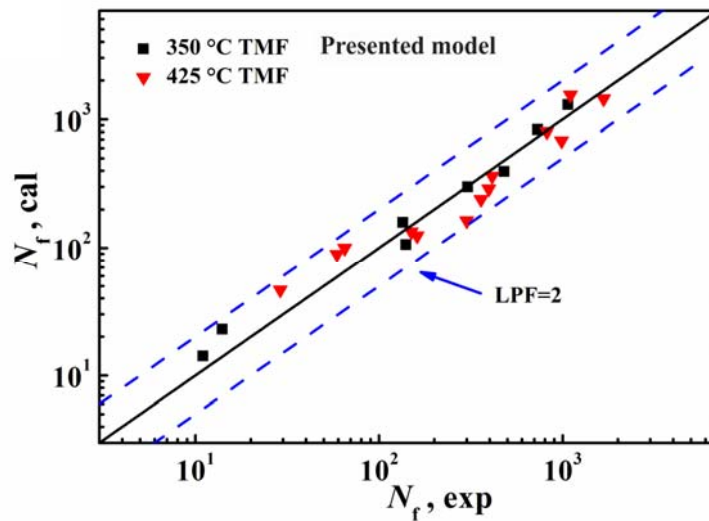
$$N_f = (W_0 / W_s)^{1/\beta}$$



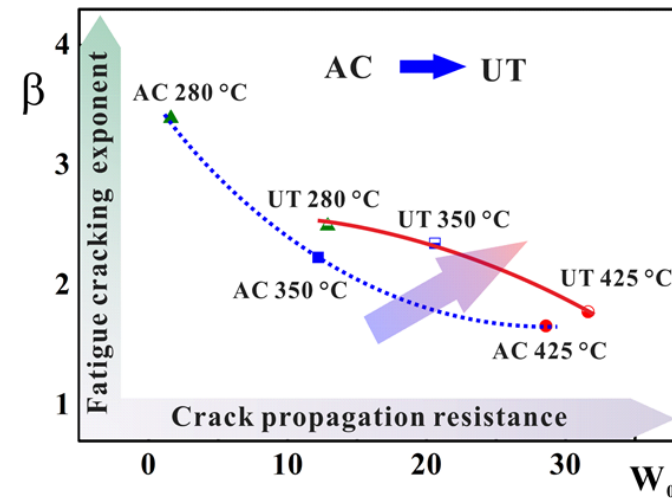
$$W_{a, TMF} = A + K \varepsilon_{mech} + W_{a, LCF}$$

$$W_s^{TMF} = W_0 \left(\frac{v_{LCF}}{v_{TMF}} \right)^{-\beta(1-k)} \cdot N_f^{-\frac{1}{\beta}}$$

Prediction model at HTs + suitable correction = TMF life prediction



High accuracy, Less test



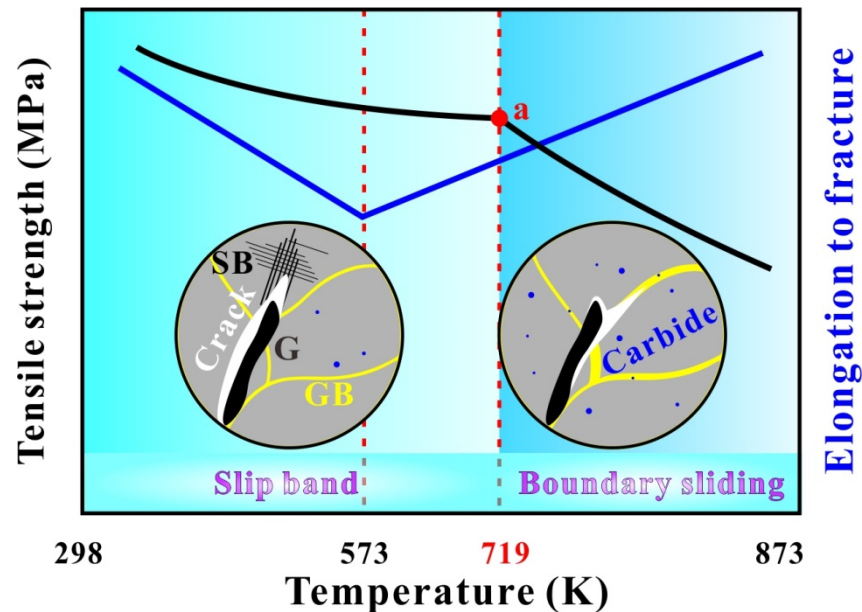
Life optimization

五、主要结论

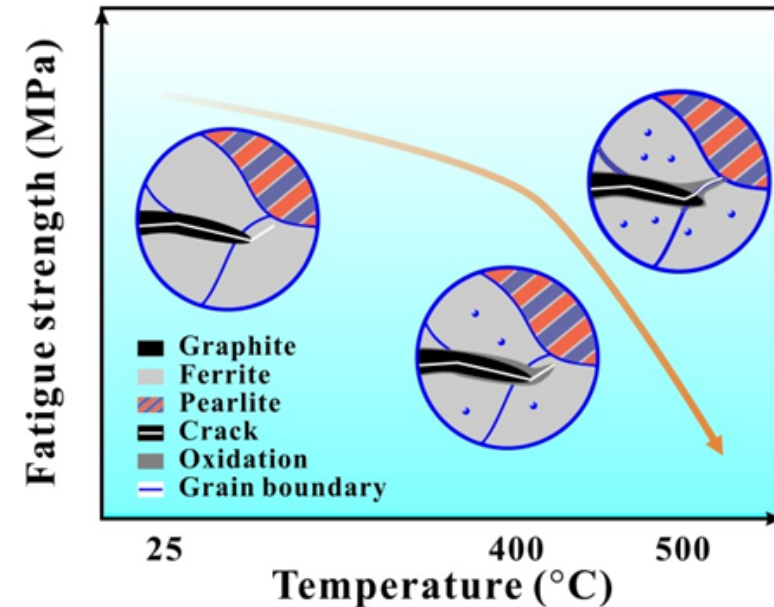
Main conclusion

Conclusion: Strength prediction at HTs

Tensile strength



Fatigue strength



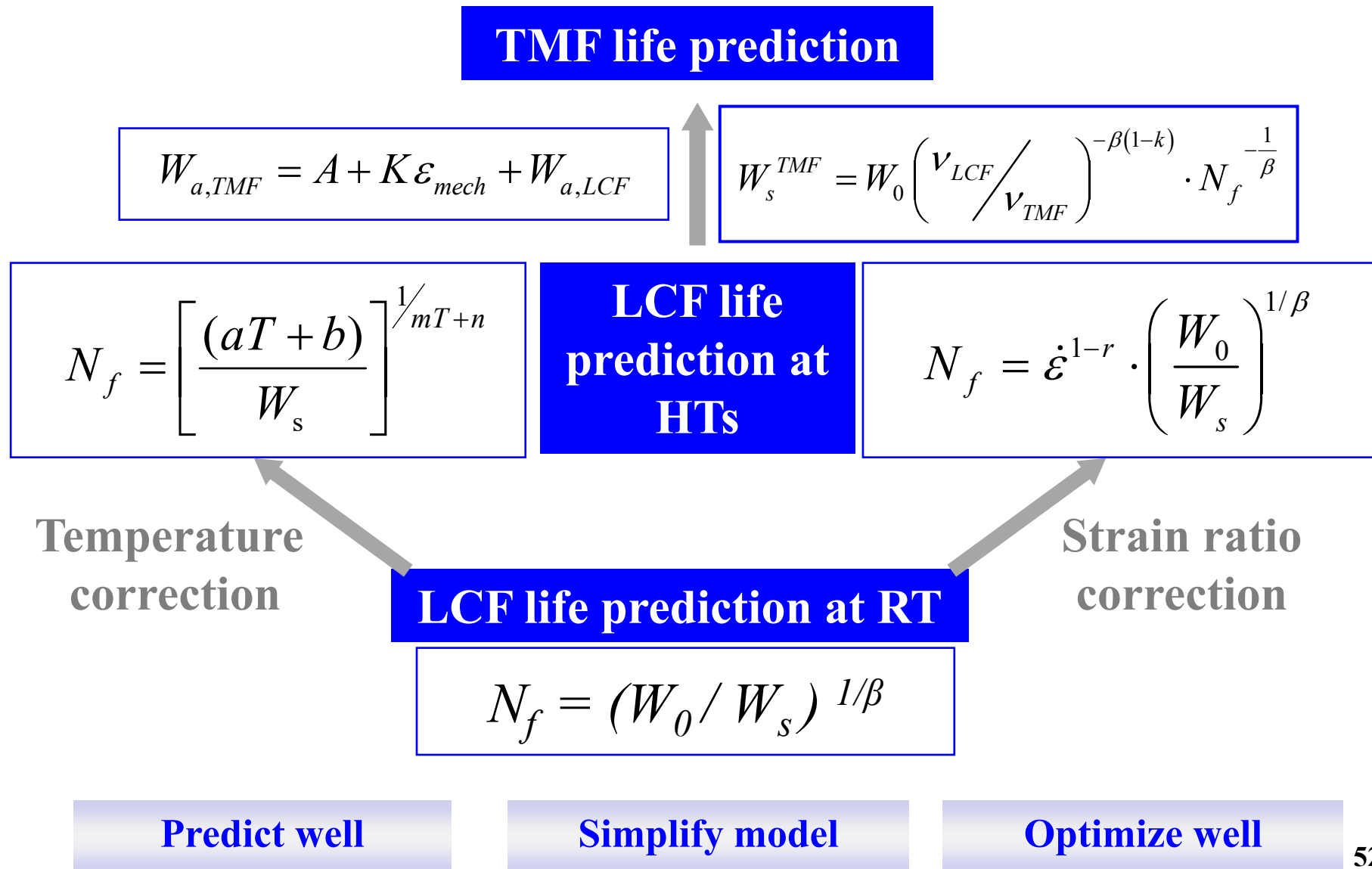
$$\sigma_b = \sigma_0 \exp(-BT) \cdot \exp(AT \ln T), T \leq 450^\circ\text{C}$$

$$\sigma_b = \sigma_0 \exp(Q/RT), T > 450^\circ\text{C}$$

$$\sigma_w = \sigma_w^f \sqrt{1 - \frac{1}{\sqrt{1 + w_f/w_g}}}$$

Based microscopic mechanism

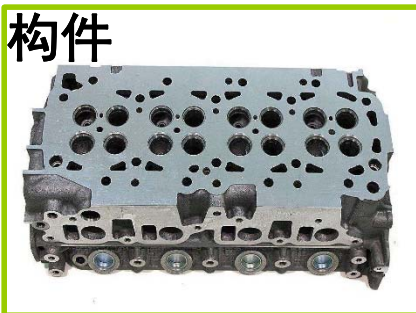
Conclusion: Life prediction at HTs



装备



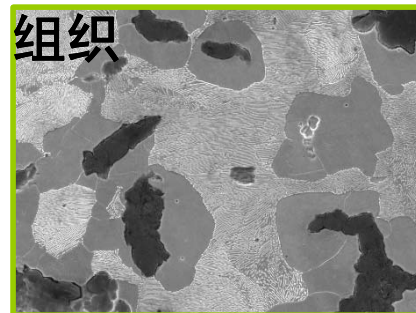
构件



材料



组织



关键共性问题理性简便解决方案



Tensile, fatigue properties, damage mechanism, temperature

1. Qiu Y, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2016; 664: 75.
2. Qiu Y, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2016; 667: 290.
3. Zhang MX, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2017; 698: 63.
4. Zhang YY, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2018;713:260.
5. Qiu Y, Pang JC*, *et al* , *Int. J. Fatigue* 2018; 112: 84.
6. Jing GX,Pang JC*, *et al* , *Eng. Fail. Anal.* 2018 ; 90: 36.
7. Qiu Y, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2018; 724: 324.
8. Zou CL, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2018; 724: 606.
9. Qiu Y, Pang JC*, *et al* , *Int. J. Fatigue* 2018; 117:450.
10. Zou CL, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2019; 763: 138101.
11. Liu QY,Pang JC*, *et al* , *Mater. Sci. Eng. A* 2019; 764: 138248.
12. Zhang MX, Pang JC*, *et al* , *Mater. Sci. Eng. A* 2020;771:138671.
13. Zou CL, Pang JC*, *et al* , *Int. J. Fatigue* 2020; 135: 105576.
14. Zhang YY, Pang JC*, *et al* , *J. Mater. Res. Technol.* 2020; 9: 7002.

Tensile, fatigue properties, damage mechanism, temperature

1. Wang M, Pang JC*, *et al*, *Mater. Sci. Eng. A* 2017; 704:480.
2. Wang M, Pang JC*, *et al*, *Mater. Sci. Eng. A* 2018; 715:62.
3. Wang M, Pang JC*, *et al*, *Adv. Eng. Mater.* 2018;20:201700610.
4. Liu HQ, Pang JC*, *et al*, *Adv. Eng. Mater.* 2018;20: 201700972.
5. Wang M, Pang JC*, *et al*, *Mater. Sci. Eng. A* 2019; 759:797.
6. Wang M, Pang JC*, *et al*, *Int. J. Fatigue* 2019; 127:268.
7. Liu HQ, Pang JC*, *et al*, *Fatigue Fract. Eng. Mater. Struct.*, 2019; 42:2372.
8. Wang M, Pang JC*, *et al*, *J. Mater. Res. Technol.* 2019; 8:4556.
9. Liu HQ, Pang JC*, *et al*, *Mater. Charact.* 2020; 159:110032.
10. Wang M, Pang JC*, *et al*, *J. Mater. Res. Technol.* 2019; 8:4556.
11. Wang M, Pang JC*, *et al*, *Mater. Sci. Eng. A.* 2020; 783: 139279.

Representative achievement

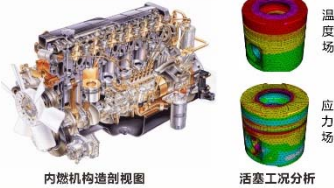


IMR 活塞铝硅合金损伤机制与使役性能预测

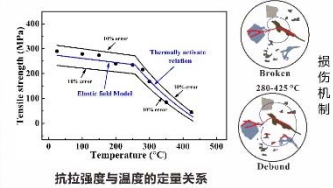
庞建超、王 猛、刘海全、刘 睿、马 荻、张 鹏、李守新、张哲峰

活塞构件长期在循环应力、温度的环境中工作，导致构件不同部位经常发生疲劳破坏。本研究选择典型铝硅合金系统研究拉伸、高周疲劳、低周疲劳和热机械疲劳性能与损伤机制，分别建立了抗拉强度、疲劳强度和疲劳寿命与温度的定量关系；建立铸造缺陷与疲劳强度的定量关系；基于损伤机制和能量法，考虑应变率的影响，提出通过少量低周疲劳预测热机械疲劳寿命的简便方法；搭建高温材料与结构疲劳寿命预测系统。这些研究成果对内燃机活塞构件服役安全和性能优化提供重要的技术支持。

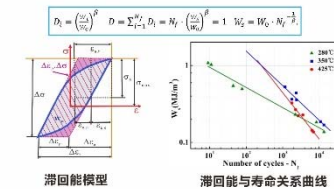
一、内燃机的构造与活塞服役工况



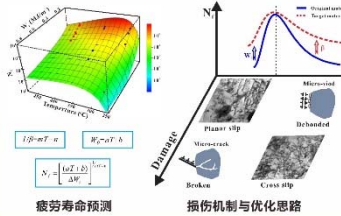
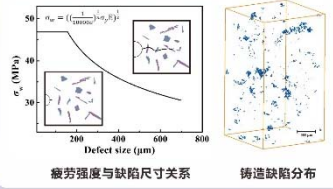
二、不同温度拉伸性能与损伤机制



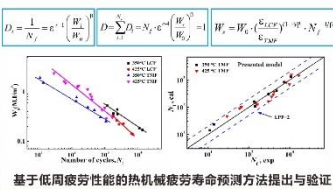
三、低周疲劳性能预测方法与机制



四、高周疲劳性能预测与开裂机制



五、简便热机械疲劳寿命预测方法



- [1] Wang M, et al. Mater. Sci. Eng. A 2017; 704:480.
 [2] Wang M, et al. Adv. Eng. Mater. 2018; 20: 201700610.
 [3] Wang M, et al. Mater. Sci. Eng. A 2019; 759: 797.
 [4] Liu HQ, et al. Adv. Eng. Mater. 2018; 20: 201700972.
 [5] 庞建超等, 材料高温材料与结构疲劳寿命预测系统V1.0
 [6] Wang M, et al. Mater. Sci. Eng. A 2018; 715:62.
 [7] Liu HQ, et al. Adv. Eng. Mater. 2018; 20: 201700972.
 [8] 庞建超等, 材料高温材料与结构疲劳寿命预测系统V1.0

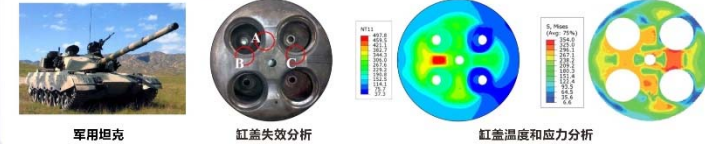
联系人：庞建超, 024-83978226, jcpang@imr.ac.cn; 张哲峰, 024-23971043, zhzhzhang@imr.ac.cn

IMR 缸盖蠕墨铸铁损伤机制与使役性能评价

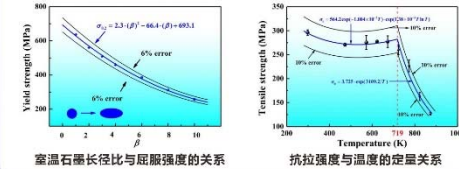
庞建超、邱 宇、邹成路、张孟泉、张艳艳、朱 刚、刘 睿、李守新、张哲峰

高功率密度柴油机服役过程中，缸盖要长期承受高强度的循环机械与热载荷，进而使鼻梁区产生疲劳失效。本研究系统分析表征蠕墨铸铁拉伸、高周疲劳、低周疲劳和热机械疲劳性能与损伤机制，分别建立了抗拉强度、疲劳强度、低周疲劳寿命与温度的定量关系；建立微观组织特征与屈服强度、疲劳强度的定量关系；基于服役工况、损伤机制和预测方法相似性原则，提出通过少量低周疲劳预测热机械疲劳寿命的简便方法。这些机制模型，为高温材料和构件疲劳性能的预测与优化提供新思路。

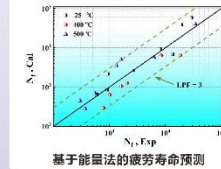
一、缸盖应用背景与服役工况



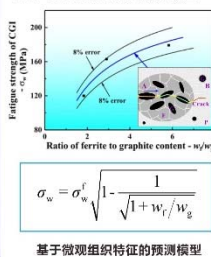
二、室温与高温拉伸性能表征



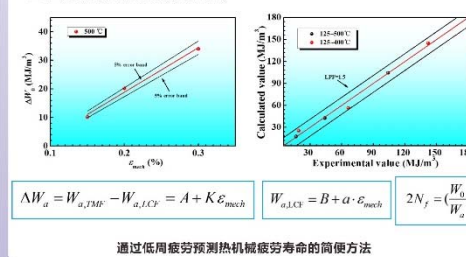
三、低周疲劳性能预测方法



四、高周疲劳性能预测模型



五、热机械疲劳寿命预测方法



- [1] Qiu Y. Mater. Sci. Eng. A 2016; 664: 75.
 [2] Qiu Y. Mater. Sci. Eng. A 2016; 667: 290.
 [3] Qiu Y. Int. J. Fatigue 2018; 112: 84.
 [4] Qiu Y. Mater. Sci. Eng. A 2018; 724: 324.
 [5] Qiu Y. Int. J. Fatigue 2018; 117: 450.
 [6] Zou CL. Mater. Sci. Eng. A 2018; 724: 606.
 [7] Zou CL. Mater. Sci. Eng. A 2019; 763: 138101.
 [8] Zhang MX. Mater. Sci. Eng. A 2017; 698: 63.
 [9] Jing GX. Eng. Fail. Anal. 2018; 90: 36.
 [10] Zhang YY. Mater. Sci. Eng. A 2018; 713: 260.
 [11] 庞建超等, 基于材料微观组织性能预测与优化的疲劳寿命预测系统V1.0

联系人：庞建超, 024-83978226, jcpang@imr.ac.cn; 张哲峰, 024-23971043, zhzhzhang@imr.ac.cn

Cast iron(CGI\SGI\FGI) cast Al(Hypoeutectic/Eutectic) Steel Cu alloy

Testing & Devolvement & Research



热机疲劳



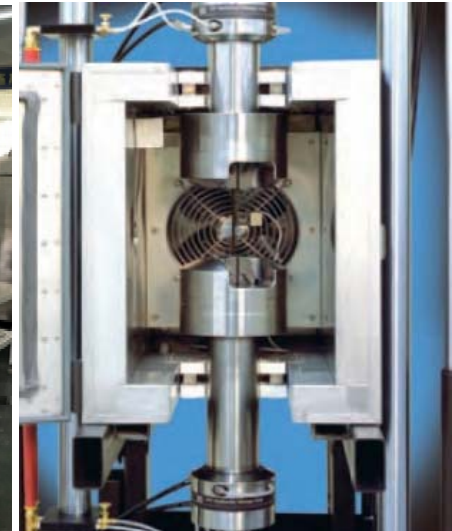
高温微缩



蠕变



热疲劳



环境疲劳



高温超高



高温硬度



高温原位

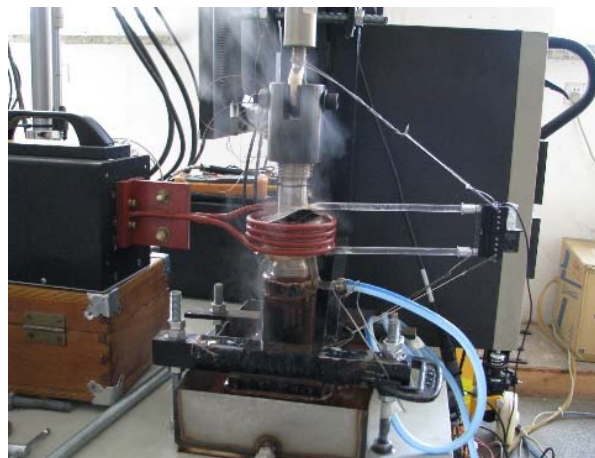


纳米压痕

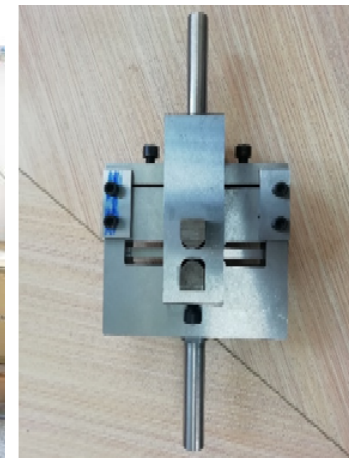
力学测试平台：40余台疲劳机



波纹管



恒载热疲劳



四点弯曲



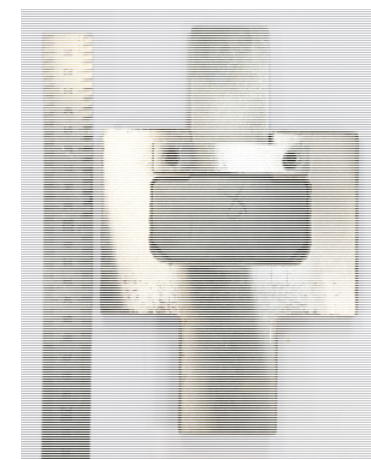
叶片热机疲劳



连杆疲劳



齿轮疲劳



磁轭疲劳

结合服役工况 定制测试方案

Basquin, 1910

$$\sigma_a = \sigma_f (2N_f)^b$$

$$D_i = \frac{1}{N_f} = 2 \cdot \left(\frac{\sigma_a}{\sigma_f} \right)^{-1/b}$$

Osttergren

$$\Delta W_T = \sigma_{\max} \cdot \Delta \varepsilon_p$$

$$N_f^\beta (\Delta W - \Delta W_0)^m = C$$

$$\sigma_{\max} \cdot \Delta \varepsilon_p \cdot N_f^\beta \cdot \nu^{\beta(k-1)} = C$$

Sehitoglu, 1989

$$\frac{1}{N_f} = \frac{1}{N_f^{\text{fat}}} + \frac{1}{N_f^{\text{ox}}} + \frac{1}{N_f^{\text{creep}}}$$

$$\frac{\Delta \varepsilon_{\text{mech}}}{2} = \frac{\sigma'_f}{E} (2N_f^{\text{fat}})^b + \varepsilon'_f (2N_f^{\text{fat}})^c$$

$$\frac{1}{N_f^{\text{ox}}} = \left[\frac{h_{cr} \delta_0}{B \Phi^{\text{ox}} K_p^{\text{eff}}} \right]^{-\frac{1}{\beta}} \frac{2(\Delta \varepsilon_{\text{mech}})^{(2/\beta+1)}}{\dot{\varepsilon}^{1-(\alpha-\beta)}}$$

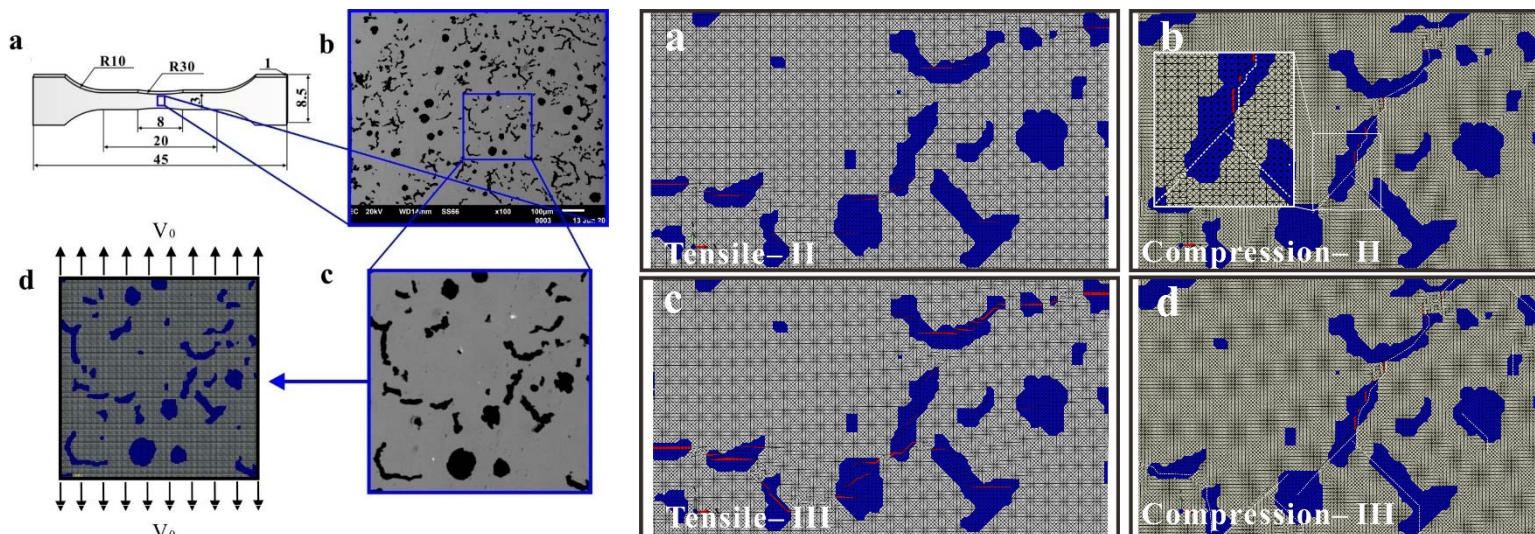
$$\frac{1}{N_f^{\text{creep}}} = \Phi^{\text{creep}} \int_0^{t_c} A \exp \left(-\frac{\Delta H}{RT(t)} \right) \cdot \left(\frac{\alpha_1 \bar{\sigma} + \alpha_2 \sigma_H}{K(T)} \right)^m dt$$

$$\Phi^{\text{creep}} = \frac{1}{t_c} \int_0^{t_c} \exp \left[\frac{1}{2} \cdot \left(\frac{(\dot{\varepsilon}_{th}/\dot{\varepsilon}_{\text{mech}}) - 1}{\xi^{\text{creep}}} \right)^2 \right] dt$$

Msc.Fatigue | FEMFAT | ANSYS® | ABAQUS

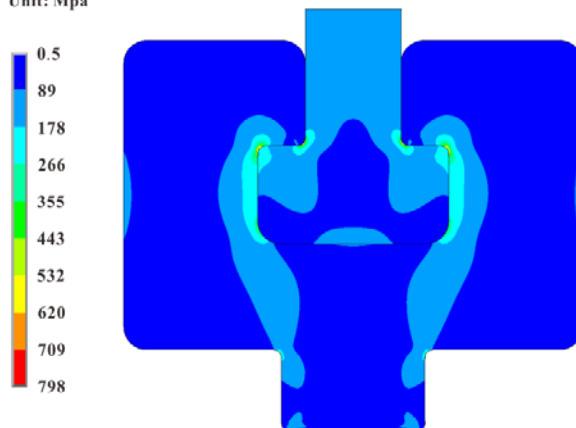
国际国家标准 FKM IIW ASME NASA GB DIN BS EN

基于组织力学性能

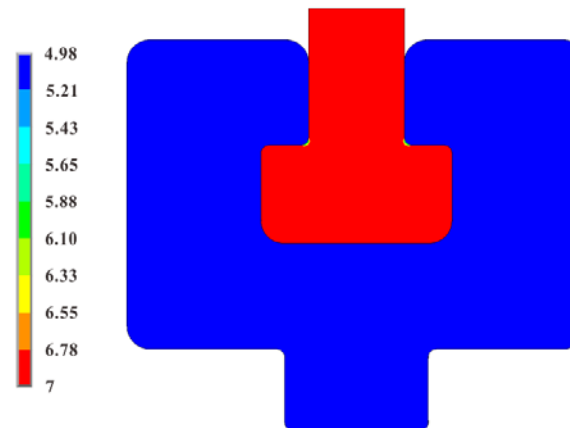


模拟构件分析

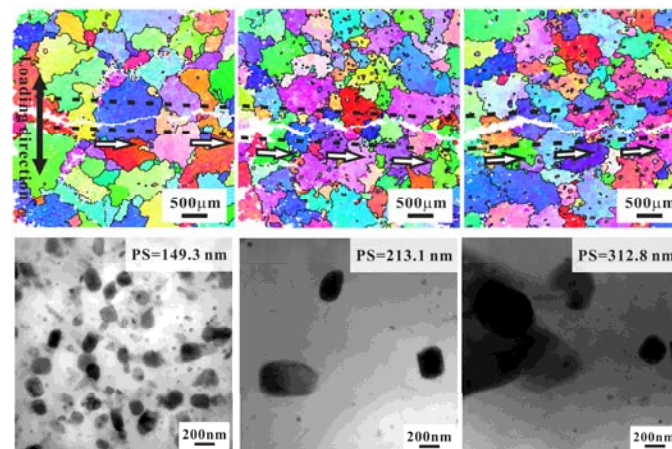
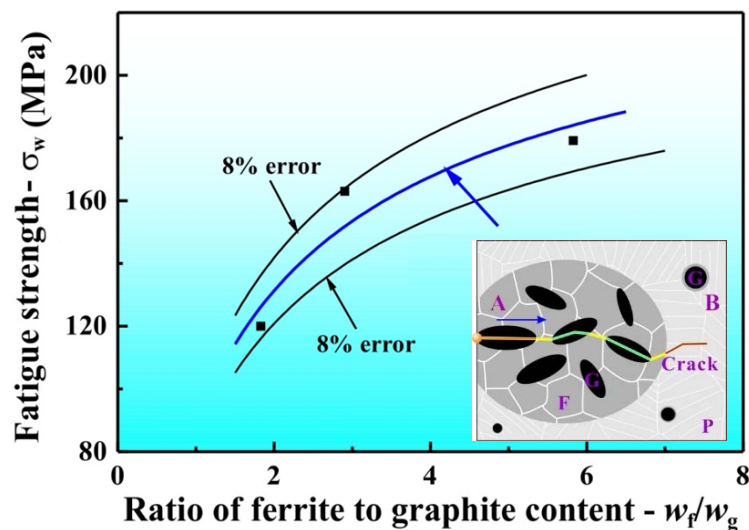
(a) Type: Equivalent (von-Mises) Stress
Unit: Mpa



(b) Type: Logarithmic fatigue life

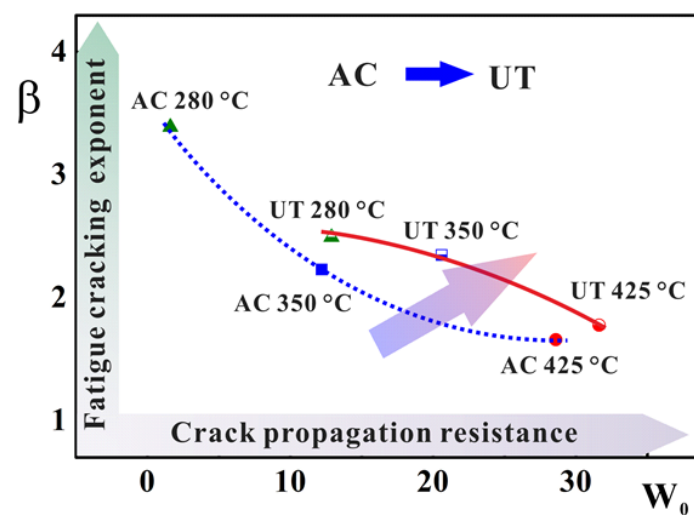
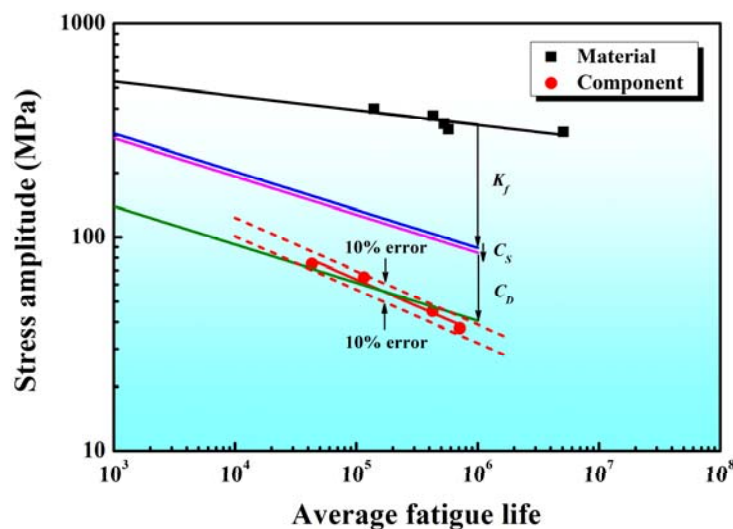


材料预测



损伤机制

规构件预测



性能优化

基于服役工况测试、分析、研究和优化方案

转子构件疲劳寿命预测

预测寿命: 38.5

项目	状态
工况条件	已输入
疲劳性能	已输入
拉伸性能	700 MPa
拟合方式	Basquin
修正系数	C_S C_D
平均应力	Goodman
预测模型	Miner

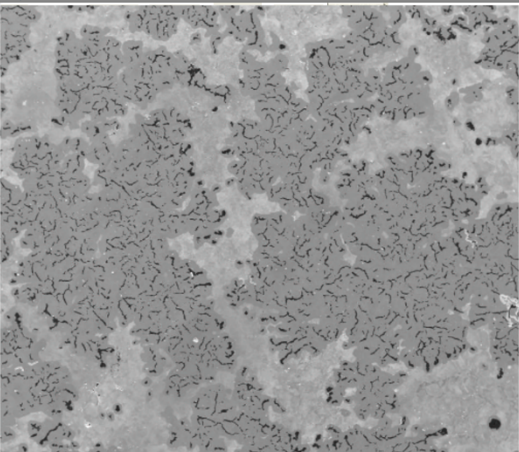
预测模型: Miner
平均应力修正: Goodman
拟合方式: Basquin
修正系数: $C_S = 0.6$
 $C_D = 0.9$

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CHINA SOUTHERN POWER GRID
哈电集团
HE
DEC
东方电气
DONGFANG ELECTRIC

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保存
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金属材料疲劳强度分析

有效区域(拉框选择)
X: 0 - 537
Y: 0 - 494
比例尺(拉线选择)
0 = 0 μm
基于基体组织 基于夹杂物尺寸
基于珠光体区裂纹长度
石墨百分比: 20
铁素体百分比: 42
珠光体百分比: 38
计算
疲劳强度: 134.119643 (MPa)



材料力学性能数据管理

材料浏览 专利及标准浏览 个人中心 用户管理 专利及标准管理 您好, 管理员 | 退出

快速浏览 高级检索

发布日期 牌号 材料 工艺 类型 高周疲劳 成分

检索 重置

材料	牌号	工艺	成分	类型	发布日期	查看	
154	铝及铝合金	A356-T6	铸造	Si:6.66;Mg:0.38;Fe:0.15;Ni:0.01;Cr:0.00;Sn:0.00	高周疲劳,拉伸	2019-03-11	查看
155	铝及铝合金	7055	锻造-冷轧		拉伸,高周疲劳	2019-03-14	查看
156	铝及铝合金	Al 4xxx	铸造	Si:12.66;Mn:0.94;Fe:0.65;Mg:0.61	高周疲劳	2019-03-14	查看
157	铝及铝合金	7075-T6	其它	Zn:5.67;Si:0.13;Fe:0.25;Cu:1.56;Mn:0.01;Cr:0.19;Ti:0.03;V:0.01	拉伸,高周疲劳	2019-03-12	查看
158	铝及铝合金	6061-T6	其它	Si:0.52;Fe:0.2;Cu:0.2;Mn:0.09;Mg:0.95;Cr:0.23;Ti:0.02;Zn:0	拉伸,高周疲劳,冲击	2019-03-11	查看
159	铝及铝合金	AS7G06-T6	铸造	Si:7.00;Mg:0.56;Fe:0.10;Cu:0.01;Mn:0.03;Ni:0.01;Zn:0.01;Pb:0	高周疲劳,拉伸	2019-03-12	查看
160	铝及铝合金	2124T851	其它	Cu:3.80;Mg:1.20;Mn:0.40;Fe:0.30;Si:0.20;Cr:0.10;Ti:0.15;Zn:0.	高周疲劳	2019-03-13	查看
161	铝及铝合金	A357	铸造	Si:7.10;Fe:0.06;Cu:0.01;Mn:0.01;Mg:0.60;Zn:0.01;Ti:0.13;Sr:0.0	高周疲劳,拉伸,冲击	2019-03-11	查看
162	铝及铝合金	6061	铸造	Mg:1.05;Si:0.65;Fe:0.34;Cu:0.26;Zn:0.13;Ti:0.12	拉伸,高周疲劳	2019-03-14	查看

200 第1 共2页 显示1到200,共207记录

根据构件服役情况定制软件(插件)

Thank for your attention

感谢关注！ 敬请提问！ 欢迎莅临沈阳指导！



疲劳分会公众号



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